# Pulsed Bias and Its Effect on Heat-induced Degradation in GaN-based HEMTs that Incorporate Barriers Having High Aluminum Content

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**Abstract** We report on the pulsed bias operations of GaN-based High-Electron-Mobility-Transistors (HEMTs) that incorporate epitaxial barriers having high Aluminum content. These devices are known to suffer from irreversible permanent degradation of their currents at Drain biases that are higher than 12-15V. This is limiting their use nowadays in most practical applications. In this work we demonstrate with modelling and experimental data that excessive self-heating caused by LO phonons in the GaN 2DEG channel is major contributor to this degradation. We also demonstrate that one approach to alleviate this excessive self-heating is by weakening the lateral field in 2DEG channel between Drain and Source. We finally demonstrate the mechanism that limits the application of higher Drain biases to be caused by combined effect from this intense self-heating and an added bias-induced strain that appears to further strain the already strained and intensely heated GaN 2DEG and accelerate the increase of traps density in GaN. Our data further suggest that epitaxial barrier and its AlN-spacer remained virtually unaffected after prolonged application of high 45V DC bias and that most epitaxial damage appears to confine in GaN, particularly under the access region between Gate and Drain and at the Drain side in the channel under the Gate where both electric-field and the self-heating are highest. An accurate Physics-based model for these HEMTs was also developed and its simulations agreed with our experimental measurements and aided us to formulate these conclusions.

Keywords HEMT, pulsed bias, AlInN, AlN

# **1. Introduction**

Recent advancements in the growth of Nitride semiconductor epitaxies have only recently started allowing depositions of Nitride epitaxies incorporating high Aluminum content that possess high spontaneous Gallium polarizations into а Nitride (GaN) 2-Dimensional-Electron-Gas (2DEG). Record high sheet carrier concentrations  $(n_{2DEG})$  close to 2.7-3.2×10<sup>13</sup> cm<sup>-2</sup> current-densities as high as 2-3A/mm and are continuously and routinely reported nowadays from High-Electron-Mobility-Transistors (HEMTs) that incorporate these epitaxies. These epitaxies are either grown by incorporating 2-4 nm strained Aluminum Nitride (AlN) barriers atop GaN [1], or by growing these barriers thicker with traces of Indium incorporated into them to alleviate the strain. The Indium molar fractions can be anywhere between 5-15%. Both approaches were reported to produce excellent

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spontaneous polarizations into the GaN 2DEG channel with highest device current-densities [2-5]. Yet one obstacle is still impeding the use of these devices in the applications that require them to operate at high bias. It is the resulting intense self-heating from which these devices are known to suffer and that confines in their GaN 2DEG and degrades their electric performance irreversibly [6-13]. This phenomenon is characteristic to HEMT devices that have high Aluminum substance in their epitaxial barriers and high corresponding sheet carrier concentrations [6-13]. It manifests itself with a pronounced increase of the self-heating in their channel which irreversibly degrades their electric performance. This intense self-heating was attributed to the continuous emissions of Longitudinal-Optic (LO) phonons when high fields are applied to the GaN 2DEG channel between the Drain and Source. LO phonons are lattice vibrations that are quantized at specific wavelengths and caused by electrons emitting a corresponding energy they lose from scattering [6-13]. The time its takes an LO phonon to convert to other phonon modes that can more easily sink out the heat from the GaN 2DEG channel is measured with the phonon-lidetime. This is inherintly high in HEMTs that have high  $n_{2DEG}$ [6-13]. Though prior works pointed to the longer

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phonon-lifetime in AlInN-based HEMTs as possible mechanism that is permanently degrading their electric performance none demonstrated tangibly how the phonon-lifetime modulates the self-heating in these devices and none explained why the high applied fields between Drain and Source drastically intensify this self-heating and accelerate the irreversible degradation of these devices. The work in [9] pointed to thermal breakdown in contacts and to a Gate sinking to result from this intense self-heating. The work in [10] reported on a degraded channel conductance in these devices due to high applied DC bias to their Drain. It explained this degradation to an increase of the traps density in GaN 2DEG from hot-electrons. Neither work did however provide a quantitative measure on the contribution from LO phonons to this intense self-heating and nor on the potentials to improve the electric performance of these devices with pulsed bias. Other work reported on the capability of these devices to operate at temperatures as high as 1300<sup>o</sup>K when high thermal conductive substrates are incorporated [14]. Same work showed that current-densities as high as 3A/mm can be reached when these devices are cooled down to cryogenic temperatures. These are highest current-densities ever reported from a solid-state transistor which therefore implies that if these transistors are integrated in miniaturized radar units for next generation fighter-jets they can deliver highest RF powers that will enable detection of enemy jets at distances three times larger than what today's state-of-the-art AlGaN/GaN HEMTs can deliver (given that the typical current-densities in today's AlGaN/GaN HEMTs are close 1A/mm). Such dramatic enhancement in performance can however come only if this irreversible permanent degradation of the electric performance of these devices is adequately tackled. In this work we demonstrate for the first time that heat that generates in GaN 2DEG due to electron scattering with other than LO phonons may be suppressed when low duty-cycle pulsed-bias is utilized whereas the intense heat that generates due to electron scattering with LO phonons tends to remain confined in GaN 2DEG and can still limit the electric performance of these devices. Aided with accurate simulations that precisely duplicated our devices Current-Voltage characteristics (I-V's) we extracted their corresponding channel temperatures at which their irreversible permanent degradations started to incur. We observed that these devices degraded irreversibly as the intense heating in their GaN 2DEG channel approached the thermal limits for bulk GaN (~  $2500^{\circ}$ K). We also observed that inverse-piezoelectric effect can accelerate the damage to an already intensely heated GaN 2DEG when higher potentials are applied between Drain and Gate. We further demonstrate with simulations that one approach to alleviating this intense self-heating caused by electron scattering with LO phonons and that confines in the GaN 2DEG channel is by weakening the lateral field in this channel.

## **2. Device Structure**

The epitaxial layers for HEMT devices were grown in Aixtron 2000HT reactor using standard precursors on

insulating 4H-SiC substrates. A 0.3µm thick AlN buffer was grown first over SiC substrates. This was then followed with growth of 1.5µm thick semi-insulating GaN. Growth of Ultra-thin (2-3nm) AlN spacer and ~9nm thick AlInN barrier were carried next. Purpose of this AlN spacer is to alleviate the channel scattering at the GaN interface. The Aluminum molar composition in AlInN barrier was targeted around 90%. A 20Å AlN cap layer was finally deposited with pulsed-Atomic-Layer-Epitaxy (PALE) to crown the AlInN barrier. This AlN cap was demonstrated to further stretch the Polarization effects and alleviate the surface Breakdown [15]. Devices were fabricated with relatively long 1.25µm Gates (LG) that still enabled a short-channel device performance, and with wide Gate-to-Source and Gate-to-Drain openings (lgs and lgd). Reasons for these wider openings are to weaken the lateral field between Drain and Source and to lessen the phonon-induced self-heating. The Gate width was same on all fabricated devices equaling 8×50µm. The Gate-to-Source openings were 2µm wide and the Gate-to-Drain openings were between 6µm and 9µm wide in different devices. Total of 12 devices were used in this study from which 3 had an  $lgd = 9\mu m$ . The measured data in this manuscript figures represent typical performances in devices having same lad. I-V's of all devices having a same lad were matched within a 5-15% error. Ti/Al/Ti/Au multilayer Ohmic contacts were deposited by e-beam evaporation followed by a 30s rapid thermal annealing at 850°C in N<sub>2</sub> atmosphere. A Ni/Au bilayer was deposited for the Gate electrode followed by a 700Å thick PECVD Si<sub>3</sub>N<sub>4</sub> layer for device passivation. 4 other similar devices incorporating plain AlN barriers that are close to 2-2.5nm thick were also fabricated with PALE. The average measured Hall Mobility  $(\mu h)$  and sheet carrier density (nsh) on all devices were close to  $1300 \text{ cm}^2/(\text{V.s})$  and  $2.85 \times 10^{13} \text{ cm}^{-2}$  respectively. "Figure 1" shows cartoon schematics for the fabricated device structures.



**Figure 1.** Cartoon schematics of our HEMTs that incorporate plain AlN barriers and those that incorporate thicker barriers containing close to 10% Indium

## 3. Device Model and Results

Though the devices having plain AlN barriers exhibited a similar trend to self-heating as the devices that incorporated AlInN barriers with I0% Indium, in order not to crowd our figures we show in this work the electric characteristics and the self-heating in the devices that incorporated AlInN barriers only. The conclusions are however similar for the devices incorporating plain AlN barriers.

#### 3.1. Device Model

The short-channel saturation-velocity model for field-effect transistors [16] was used for simulating the I-V curves of devices. These simulations accounted for the dependence of electron saturation-velocity and electron channel mobility on the channel temperature. We modelled the dependence of mobility on channel temperature due to electron scattering with LO phonons ( $\mu LO$ ) following the Bose-Einstein distribution function in [9]. This is since it is well known and it was specifically demonstrated in [9, 17] that the temperature rise in GaN 2DEG due to excited LO phonons follows a Boltzmann-like distribution which consequently causes  $\mu LO$  to follow a same distribution with

$$\mu LO(T) = \frac{\kappa}{NB(T)} \tag{1}$$

NB(T) is the Bose-Einstein distribution function as it was defined in [9] and K is a constant. This mobility adds in parallel to another mobility ( $\mu LT$ ) that results from other mechanisms of electron scattering.

$$\frac{1}{\mu n(T)} = \frac{1}{\mu LO(T)} + \frac{1}{\mu LT}$$

 $\mu LT$  includes scattering's that are less temperature-dependant such as those due to interface roughness, defects, piezoelectric scattering and electron-electron scattering. The expression for NB(T)from [9, 17] is

$$NB(T) = \frac{1}{e^{(h \times f/(KB \times T))} - 1}$$
(2)

Similarly, our temperature-dependent saturation-velocity model was fitted to follow experimental measurements in [10] for saturation-currents at temperature up to 1300<sup>o</sup>K. This empirical fit is

$$vsat(T) = vsat \times 300^{0.531} \times T^{-0.531}$$
 (3)

*vsat* is the desired value for vsat(T) at 300°K.

The model for device short-channel current conforms to the equation,

$$I = \frac{gch \times VD}{\left(1 + \left(\frac{VD}{Vknee}\right)^{\rho}\right)^{1/\rho}}$$
(4)

*VD* is the voltage to drain,  $\rho$  is a fitting parameter that models the transition of current from its linear-mode to saturation (we used  $\rho = 10$  in our simulations), and *gch* is its extrinsic linear channel conductance expressed with

$$gch = \frac{gchi}{1 + gchi \times Rs + Rgd} \tag{5}$$

$$gchi \cong WG \times q \times n_{2DEG} \times \mu n(T)/LG \qquad (6)$$

 $q \times n_{2DEG}$  is the 2DEG sheet charge estimated with:  $C_{barrier} \times (VGS - VT)$ . VT is the device threshold-voltage,  $C_{barrier}$  is the capacitance per unit-area of device barrier and VGS is the potential applied between Gate and Source. Rs and Rgd are the extrinsic Source and Drain resistances of the device that are function of temperature. WG and LG are the device Gate width and Gate length respectively. q is the electron charge unit  $(1.6 \times 10^{-19} \text{C})$ . Estimate for Rs and Rgd are

$$Rs \cong \frac{lgs}{q \times \mu n(T) \times WG \times nsh}$$

$$Rgd \cong \frac{lgd}{q \times \mu n(T) \times WG \times nsh}$$
(7)

*lgs* and *lgd* are the Gate-to-Source and Gate-to-Drain access regions. Estimate of the Vknee in short-channel field-effect devices is

$$Vknee(T) = v_{sat}(T) \times LG/\mu n(T)$$
(8)

It is estimated from [16] that the currents in short-channel Field-effect devices follow saturation-velocity model when

$$v_{sat}(T) \times \frac{LG}{\mu n(T)} < (VGS - VT)$$
(9)

Our devices have:  $LG = 1.25 \mu m$ , VT = -4, and a low field room temperature  $\mu_{LT} \cong 2000 \text{ cm}^2/(\text{V.s})$  and  $vsat(T) = 8.5 \times 10^7 \text{ cm/s}$ . Therefore the short-channel criterion of equation (9) is met at  $300^{\circ}$ K for VGS = +1.5V. When this criterion is not met, Vknee = VGS - VT and the saturation device current of equation (4) becomes

$$= \frac{\mu n(T)}{2 \times LG} \times WG \times q \times n_{2DEG} \times (VGS - VT)$$

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The rise of channel temperature due to electron scattering with LO phonons ( $T_{LO,phonons}$ ) is modelled to follow similar Bose-Einstein distribution function as our modelling of  $\mu LO(T)$ . This is since some electrons do emit more LO phonons from this scattering to further energize the lattice vibrations and increase the scattering and the self-heating for same bias. The number of electrons that emit in LO mode from this scattering is known to follow a similar statistical distribution as that of equation (2) [9, 17]. This continuous process of emissions and increased scattering latches a self-sustained feedback that thrusts higher emission rate for LO phonons and causes consequent higher rate of heat build-up in GaN 2DEG which slows its diffusion to substrate and forces most of it to confine instead in the GaN 2DEG channel and raises this channel temperature. The Bose-Einstein distribution function captures the average number of excited LO phonons for given applied field and an initial temperature (Ti). Electrons do still scatter due to other scattering mechanisms (with other than LO phonons) and these also increase this channel temperature  $(T_{LT})$  due to their Ohmic conduction in it. The total temperature in 2DEG channel at given applied bias is therefore

$$T = T_{LO,phonon} + T_{LT}$$
(12)

Unlike  $T_{LO,phonon}$  that is less sensitive to the thermal conductance of substrate and to device thermal package because it tends to confine in the GaN 2DEG channel even when SiC substrates are used,  $T_{LT}$  does depend highly on the thermal conductance of substrate and/or on the device thermal package. It is modelled with

$$T_{LT} = \left[ Rth \times I \times VD \times \left( 1 - e^{-\frac{ton}{\tau}} \right) + 300^{\circ}K \right]$$
(13)

*Rth* is the equivalent thermal resistance between hot GaN 2DEG and coldest reference in device structure.  $\tau$  is the thermal time-constant for heat rise in GaN 2DEG and  $t_{on}$  is the time duration of the applied potential to Drain. When device operates with a pulsed bias ( $VD = V_{Pulse}$ ), both

pulsed and time-averaged heating caused by electron scattering with other than LO phonons can be suppressed. The pulsed heating can be suppressed with  $t_{on} << \tau$  while a low  $t_{on}$  and a low duty-cycle suppress the time-averaged heating. Heating that is however caused due to electron scattering with LO phonons is more difficult to suppress with pulsed bias as it tends to confine in GaN 2DEG during the  $t_{on}$ . Both initial temperature (*Ti*) prior to an electric field is pulsed and the number of excited LO phonons due to this field contribute together to an increase of  $T_{LO,phonons}$ . Therefore,  $T_{LO,phonons}$  may also be reduced to lesser extent with very low duty-cycle since *Ti* will reduce and resets closer to  $300^{0}$ K at beginning of each pulse following

$$Ti = (T_{LT} - 300^{\circ} \text{K} + T_{LO,phonons}) \times e^{-(\frac{Ip-t_{on}}{\tau})} + 300^{\circ} \text{K}$$

Tp is the period of pulsed signal. The Bose-Einstein relation between  $T_{LO,phonons}$  and the total average number of excited LO phonons as it was described in [9, 17] is

$$n_{LO,ex}(VD) + n_{LO}(Ti) = \frac{n_{LO,max}}{e^{\left(\frac{h \times f}{KB \times T_{LO,phonons}}\right)_{-1}}}$$
(14)

 $T_{LO,phonons}$  is extracted from the above equation for the values of  $n_{LO}(Ti)$  and  $n_{LO,ex}(VD)$ .

 $n_{LO,ex}(VD)$  is the average number of LO phonons excited due to an applied field, and  $n_{LO}(Ti)$  is the average number of LO phonons at temperature Ti and that also follows a Bose-Einstein distribution function described in [9, 17] with

$$n_{LO}(Ti) = \frac{n_{LO,max}}{e^{\left(\frac{h \times f}{KB \times Ti}\right)_{-1}}}$$
(15)

*h* is the Planck's constant ( $h = 6.63 \times 10^{-34}$ J.s). *KB* is the Boltzmann's constant ( $KB = 1.38 \times 10^{-23}$  J/<sup>o</sup>K).  $h \times f$  is the quantized energy of LO phonons that are stimulated from electrons emitting upon scattering (*f* corresponds to longer infrared wavelength. We used 2.5 *THz* in our simulations).  $n_{LO,max}$  in equations (14) and (15) represents the maximum amount of available LO phonons in the volume that is covered by the GaN 2DEG (*V2DEG*).  $n_{LO,max}$  was modelled in [9, 18] with

$$n_{LO,max} = 2 \times \frac{V2DEG}{VunitCell}$$
(16)

The factor 2 in the above equation arises from the fact that two LO phonons can be available per unit cell [18]. An estimate for V2DEG from [9, 18] is

$$V2DEG = L2DEG/n_{2DEG} \tag{17}$$

L2DEG is the total length of 2DEG GaN (in our devices (L2DEG = LG + lgs + lgd)). Vunitcell is the enclosed volume in GaN Wuritzite unit cell.

$$Vunitcell = \frac{\sqrt{3}}{2} \times a^2 \times c \tag{18}$$

*a* and *c* are the lattice constants in GaN Wuritzite unit cell (a = 3.189 Å, and c = 5.18 Å). An expression for  $n_{LO,ex}$  can now be estimated [9] with

$$n_{LO,ex} = L2DEG \times \sqrt{n_{2DEG}} \times \frac{L2DEG}{LE}$$
(19)

First multiplicand in the above equation represents the total number of electrons in the 2DEG GaN channel and the

second multiplicand is the number of excited LO phonons per electron in channel. *LE* is the distance over which an electron must travel to reach an LO phonon energy equalling  $h \times f$ . It was derived in [9] to be close to

$$LE = \frac{2 \times m^* \times vsat^2(T) \times (v_{sat}(T) \times (vD/L2DEG))}{q \times (vD/L2DEG)^2 \times \mu n(T)}$$
(20)

We estimated in equation (20) an averaged value for *LE* that corresponds to an averaged magnitude for electric field  $(E_{avg})$  in entire *L2DEG* such

$$E_l \times d(l) = E_{avg} \times L2DEG = VD$$

 $E_l$  is the varying lateral electric-field through L2DEGand  $m^*$  is an estimate for the effective-electron mass when an electron reaches LE at high fields and emits energy  $h \times f$ . Its expression is  $m^* = \frac{h \times f}{vsat^2(T) \times 2}$  when assuming the electron reaches a peak velocity between scatterings that is twice its saturation-velocity (this is since saturation-velocity is the time-averaged velocity).

Our model is more accurate than that demonstrated in [9] and it differs from it by accounting to an added contribution to channel temperature due to the Ohmic conduction of electrons that scatter with other than LO phonons and to its transient rise in the channel. It also provides accurate modelling for extrinsic channel transconductance, for short-channel and long-channel *Vknee*, for device drive current and for its dependence on the channel temperature. It also accounts to the impacts from *lgd* and *lgs* on the average number of excited LO phonons and on the corresponding shift in channel temperature that these cause.

 $Rth = 75^{\circ}$ K/W, and Rth = 50K/W were used for simulating the HEMTs having 6µm and 9µm Gate-to-Drain spacing respectively. These were estimated using the nominal 4.9W/(cm.°K) thermal conductivity of a SiC substrate with oversimplified estimations that mirrored the effects from this substrate thickness (  $th_{SiC} = 220 \mu m$  ), WG and lgd on  $Rth \left(Rth = (10^{-2}/4.9) \times (th_{sic}/(WG \times lgd))\right).$ Our simulations also used  $\tau = 0.1ms$ . This was taken from simulated estimates for the temperature rise in a GaN 2DEG formed on top of high thermal conductive SiC substrate [19]. We also used k = 4 in all our simulations, and  $\mu LT = 2000 \text{ cm}^2/(\text{V.s}).$ 

#### 3.2. Results from Measurements and Simulations

Pulsed measurements were carried on all fabricated devices. The Magnitude of the voltage pulse to their Drain  $(V_{Pulse})$  was set to rise from 0V and remain constant during time interval:  $t_{on}$ . The time between pulses:  $t_{off}$  was set to 1s and  $V_{Pulse}$  increased 1mV after each period Tp ( $Tp = t_{on} + t_{off}$ ). The applied potentials to Gates (*VGS*) were synchronized to  $V_{Pulse}$ . Unlike in conventional 60Hz curve-tracers that capture effects of averaged self-heating in response to 60Hz sweeps to device terminal, our measurements used the Accent-DIVA-D265 system that sampled magnitudes for the device current at the end of every  $t_{on}$ . This permitted the capture of effects from

self-heating during each time-interval in which different  $V_{Pulse}$  was applied. This consequently mirrored the real-time dynamic response of the currents of devices due to an increasing  $V_{Pulse}$ . A schematic illustrating the time-dependence of  $V_{Pulse}$  as it was used in our measurements is shown in "Figure 2".



Figure 2. Schematic illustrating the time-dependence of  $V_{Pulse}$ 

Data in "Figure 3a" show the corresponding measured I-V characteristics on two distinct devices that have same widths and lengths ( $WG = 400 \mu m$  and  $LG = 1.25 \mu m$ ) and that were biased with VGS = 1.5V.  $t_{on}$  was set to 0.61ms in measuring one device (HEMT-A) and to 10µs in measuring the other device (HEMT-B). lgd was 6µm in HEMT-A and 9µm in HEMT-B. To quantify the self-heating in these devices we picked an arbitrary measure from their measured I-V's using metric:  $|\Delta I/(Imax \times \Delta V)|$ . This captured for given  $t_{on}$  and VGS the heat-induced reversible decrease of their saturation-currents  $\Delta I$  with the increase of  $V_{Pulse}$ relative to the peak value of their saturation-current Imax. The self-heating of both devices was also simulated following the model described in previous section. It was observed that their measured I-V's were fully reversible and recoverable for magnitudes of V<sub>Pulse</sub> below 10.5V for HEMT-A, and below 20V for HEMT-B, and provided that  $V_{Pulse}$  never exceeded these cut-offs prior on the corresponding device. It was further observed that as  $V_{Pulse}$ exceeded these cut-offs (that correspond to the marks B and B' in "Figure 3a") the currents of both devices ceased to follow their smooth steady trend that simulations showed and both currents started an unexpected uptick at these cut-offs. The root-cause of these upticks as it will be demonstrated later are caused by continuous shifts of their phonon-lifetime that continuously reduced the self-heating in these devices as  $V_{Pulse}$  continued to increase and further degrade their  $n_{2DEG}$ . Measurements that extracted their  $n_{2DEG}$  as  $V_{Pulse}$  was increasing indicated that a permanent irreversible damage did incur to both devices after  $V_{Pulse}$ reached its corresponding cut-off on each device.

This was confirmed from measurements of the Gate capacitance per unit-area on both devices for each magnitude of  $V_{Pulse}$  from which a corresponding  $n_{2DEG}$  was calculated. These showed clear gradual irreversible degradations after  $V_{Pulse}$  reached its corresponding cut-off on each device. After  $V_{Pulse}$  reached 10.5V,  $n_{2DEG}$  started reducing permanently in HEMT-A, from its initial measured magnitude of  $1.9 \times 10^{17} cm^{-2}$  to a value of  $1.4 \times 10^{13} cm^{-2}$  at  $V_{Pulse} = 20V \cdot n_{2DEG}$  started to similarly reduce permanently in HEMT-B after  $V_{Pulse}$  reached 20V

on this device, from same initial magnitude of  $1.9 \times 10^{13} cm^{-2}$  to  $1.15 \times 10^{13} cm^{-2}$  at  $V_{Pulse} = 25V$ . These values for  $n_{2DEG}$  were estimated with



**Figure 3a.** Measured and simulated dynamic currents response to  $V_{Pulse}$  for HEMT-A and for HEMT-B. *VGS* =1.5V. (Devices that incorporated plain AlN barriers showed a similar trend)



**Figure 3b.** Measured  $IG_{Pulse}$  and electrically extracted  $n_{2DEG}$  versus  $V_{Pulse}$  on the corresponding HEMT-A and HEMT-B of "Figure 3a". (Devices that incorporated plain AlN barriers showed a similar trend)

 $C_{AlInN}$  is the Gate-to-Source capacitance. It was measured with Agilent 4284A C-V meter while VGS = 1.5V. "Figure 3b" shows the corresponding measured pulsed Gate leakage currents ( $IG_{Pulse}$ ) and  $n_{2DEG}$  versus  $V_{Pulse}$  for the HEMTs of "Figure 3a". Apparently no correlation exists in both devices between the onsets of their degraded  $n_{2DEG}$  that "figure 3a" shows and the upticks of *IG<sub>Pulse</sub>* that "Figure 3b" shows. The onsets for their degraded  $n_{2DEG}$  in "Figure 3b" however matched precisely the magnitudes for  $V_{Pulse}$  in "Figure 3a" at which the currents of both devices started deviating from their expected and simulated trends (corresponding to the marks *B* and *B*' in both figures). We also observe from "Figure 3a" that our measure of the self-heating in devices that metric  $|\Delta I/(Imax \times \Delta V)|$ captures is less pronounced in HEMT-B that was pulsed with smaller  $t_{on} = 10 \mu s$  (compared to the larger  $t_{on} = 0.61 m s$ used to measure HEMT-A).  $|\Delta I / (Imax \times \Delta V)|$  was 60mS for HEMT-A but only 23mS for HEMT-B.

A re-sweep (or a second sweep) of  $V_{Pulse}$  on HEMT-A with  $V_{Pulse} < 20V$  and same  $t_{on} = 0.61$ ms confirmed a clear

degraded device performance that manifested itself with reduced peak saturation-current and higher Vknee (the Vknee increased from 5V to 10V). This re-sweep is also shown in "Figure 3a". This observed irreversible (permanent) increase of the Vknee confirms that substantial damage had incurred to the GaN 2DEG channel below the access region between Drain and Gate. The self-heating on same device however reduced drastically as is demonstrated from the measured  $|\Delta I/(Imax \times \Delta V)|$  on same device during this re-sweep (it reduced to 20mS from its 60mS high when the device was initially measured. The cause of this reduction of self-heating after  $n_{2DEG}$  degraded is same as what caused the currents of HEMT-A and HEMT-B in same figure to begin increasing once their  $n_{2DEG}$  started continuously degrading with increasing  $V_{Pulse}$  and is attributed to a reduced phonon-lifetime. A second re-sweep for same range of  $V_{Pulse}$  on HEMT-A confirmed same I-V as that of first re-sweep that is shown in "Figure 3a". This implies that once device degrades at given Drain potential and VGS no further degradations incur to device provided that no higher bias gets applied to its Drain.

Because the onsets for the irreversible degradations of  $n_{2DEG}$  in both devices (that "Figure 3b" shows) incurred as the self-heating of their currents (that "Figure 3a" shows) were continuously and gradually increasing; and because HEMT-A that was measured with larger  $t_{on} = 0.61$ ms and showed consequent more pronounced self-heating started degrading at lower  $V_{Pulse} = 10.5$ V; and because the  $IG_{Pulse}$ of both devices did not correlate to the onsets of their degraded  $n_{2DEG}$ , we conclude that that the observed degradations of  $n_{2DEG}$  in "Figure 3b" are predominantly caused by excessive self-heating that confines in the 2DEG channel of these devices. This conclusion is consistent with findings in [6, 10, 12] that pointed to self-heating and hot electrons in GaN 2DEG as the most likely cause to this irreversible permanent degradation of the electric performance in similar AlInN-based HEMTs.

"Figure 4" shows simulated 2DEG channel temperatures for the devices which data were shown in "Figure 3a" and "Figure 3b". Apparently the 2DEG channel temperature in HEMT-A reached magnitude at mark *B* ( $V_{Pulse} \approx 10.5V$ ) equalling

# $T_{LO,phonons} + T_{LT} \cong 2374^0 K$

 $(T_{LT} = 1607^{0}$ K, and  $T_{LO,phonons} = 767^{0}$ K). This is close to the thermal limit of Bulk GaN that is close to 2500°K. Similarly the 2DEG channel temperature in HEMT-B reached magnitude at mark *B*' equalling:  $T_{LO,phonons} + T_{LT} = 1827^{0}K(T_{LT} = 531^{0}$ K;  $T_{LO,phonons} = 1296^{0}$ K). This is also substantially high given that  $T_{LT}$  was effectively suppressed due to much lower  $t_{on} = 10\mu s$  (compared to  $t_{on} = 0.61ms$  in measuring HEMT-A). This suppression is because the thermal transient rise of channel temperature that is caused by heat that generates due to electron scattering with other than LO phonons is suppressible with high thermally conductive substrates and/or an efficient thermal device package.

The simulated temperatures in "Figure 4" are deemed accurate as they enabled same model to duplicate following

their trend with  $V_{Pulse}$  the measured I-V's of both HEMTs that "Figure 3a" shows. It is further observed from "Figure 4" that  $T_{LO,phonons}$  of HEMT-B is noticeably lower than that of HEMT-A for same range of  $V_{Pulse}$ . We believe that it is the wider  $lgd = 9\mu m$  in HEMT-B and not the lower  $t_{on} =$  $10\mu s$  while measuring this device that exerted stronger influence on lowering its  $T_{LO, phonons}$ . This is because a larger L2DEG weakens the lateral field between Drain and Source and consequently increases the length over which an electron must travel to reach its quantized energy for LO emission (following equation (20)). This in turn reduces  $n_{LO,ex}$  and consequently lowers the channel temperature due to electron scattering with LO phonons (following equations (19) and (14)). This conclusion is being further supported with the additional simulations in "Figure 5" that show the trends for  $T_{LO,phonons}$  and that for the currents of devices versus  $V_{Pulse}$  for different values of lgd and with low  $t_{on} = 10 \mu s$  that effectively suppressed  $T_{LT}$ . These simulations demonstrated clearly that larger values of lgd cause weaker lateral fields in L2DEG between Drain and Source and suppress  $T_{LO,phonons}$ .



**Figure 4.** Simulated  $T_{LO,phonons}$  and  $T_{LT}$  versus  $V_{Pulse}$  for the HEMT-A and HEMT-B of "Figure 3a" and "Figure 3b". (*VGS* =1.5V)



**Figure 5.** Simulated  $T_{LO,phonons}$  and corresponding device currents versus  $V_{Pulse}$  for different values of lgd. ( $t_{on} = 10\mu s, VGS = 1.5V$ )

Although the gradual decrease of  $n_{2DEG}$  that "Figure 3b" shows for HEMT-A and HEMT-B can be expected to cause currents of both devices to decrease, the very exact opposite was observed in "Figure 3a". We will demonstrate that these increases of currents in both devices are caused by a continuing reduction of their hot phonon-lifetime as these devices continued to degrade with the continuous increase of

 $V_{Pulse}$ . It was established in [6] that  $n_{2DEC}$  modulate the phonon-lifetime but no work demonstrated tangibly that same phonon-lifetime modulates the self-heating in GaN 2DEG and alter the devices I-V's. To demonstrating this we swept  $V_{Pulse}$  at  $t_{on} = 0.61 \,\mathrm{ms}$  and  $VGS = 1.5 \,\mathrm{V}$  on two additional new devices that were not measured prior.  $V_{Pulse}$ was swept up to 40V on one device degrading it while reducing its  $n_{2DEG}$  to  $6.8 \times 10^{12}$  cm<sup>-2</sup> and its Imax to 0.75A/mm from an initial 2.25A/mm, and it was similarly swept up to 45V on the other device degrading it while reducing its  $n_{2DEG}$  to  $5.5 \times 10^{12}$  cm<sup>-2</sup> and its Imax to 0.86A/mm from same initial magnitude. Both devices had an initial  $n_{2DEG}$  equalling  $1.9 \times 10^{13}$  cm<sup>-2</sup> prior to degrading. After these initial sweeps that irreversibly degraded both devices they were both swept again with same  $t_{on}$  and VGS and with  $V_{Pulse}$  not exceeding 27V on either. These re-sweeps are shown in "Figure 6". "Figure 6" also shows the I-V's for HEMT-A from "Figure 3a" before and after this device degraded. All currents in "Figure 6" were normalized to their Imax. Most apparent from "Figure 6" is that the self-heating in these devices that we quantified with  $|\Delta I/(Imax \times \Delta V)|$  decreased at first with decreasing  $n_{2DEG}$  and re-increased again with continuous decrease of  $n_{2DEG}$ . The  $|\Delta I/(Imax \times \Delta V)|$  reduced from 60mS in HEMT-A prior to degrading, to 20mS after  $n_{2DEG}$  degraded irreversibly on same device, to 4mS in new device which  $n_{2DEG}$  degraded more, and re-increased again to 16mS in other new device which  $n_{2DEG}$  degraded even more. Such unusual trend can only be explained with that of the phonon-lifetime that follows exact same non-linear shifts with varying  $n_{2DEG}$ . That is what actually alleviated the self-heating in the devices of "Figure 3a" once their  $n_{2DEG}$ started degrading. "Figure 6" also shows that the permanent shifts in VKnee that incurred to devices are substantial and inversely proportional to the decrease of their  $n_{2DEG}$  after they degraded from higher applied  $V_{Pulse}$ .



**Figure 6.** Measured I-V's on different devices having different  $n_{2DEG}$  and different degrees of defects in their GaN. ( $lgd = 6\mu m$ ; VGS = 1.5V;  $t_{on} = 0.61$ ms)

This confirms our earlier finding that most damage to GaN appears to have incurred below the access region between Gate and Drain and at the Drain side in the channel under the

Gate where both electric-field and self-heating are highest. "Figure 7" shows how our measured trend in "Figure 6" for  $|\Delta I/(Imax \times \Delta V)|$  versus  $n_{2DEG}$  correlates perfectly to that of the phonon-lifetime. The phonon-lifetime  $(\tau)$  in "Figure 7" was fitted following its analytic model in [6] and for the estimate of resonant  $n_{2DEG}(n_{Res2DEG})$  that was extracted from the measurements of "Figure 6".  $n_{Res2DEG}$  is the GaN 2DEG carrier concentration at which the frequency of LO phonons approaches that of hot electrons and the phonon-lifetime reaches minima. From our measured data in "Figure 6" this closely corresponds to lowest  $|\Delta I/(Imax \times$  $\Delta V$  = 4mS which corresponds to the device having an  $n_{2DEG} = 6.8 \times 10^{12} cm^{-2}$ . This therefore makes  $n_{Res2DEG}$ for our processed devices close to this value. Work in [6] had attempted to correlate the currents drop in AlInN-based HEMTs after their  $n_{2DEG}$  degraded to their phonon-lifetime after degradation but have obtained a weak and a deluded correlation. This is because the phonon-lifetime correlates to the channel self-heating and not to the degree of current drop after degradation incurs; currents do re-increase after degrading as it was specifically shown in our "Figure 3a". "Figure 8" shows the simulated trends for the channel temperatures of the devices in "Figure 6" with increasing  $V_{Pulse}$ . The measured 2 new devices that were stressed with far higher V<sub>Pulse</sub> reaching 40V and 45V had very close Imax values after they degraded (0.75 and 0.86A/mm). This consequently caused the 2nd measured new device which  $n_{2DEG}$  degraded most and that has higher corresponding rate of decrease of its current with increase of  $V_{Pulse}$  in agreement with its phonon-lifetime in "Figure 7", to have higher channel temperature at V<sub>Pulse</sub> close or equal to 20-25V. The Imax on the other hand of HEMT-A after this device degraded remained relatively high (2.03A/mm) and given that its phonon-lifetime in 'Figure 7" collapsed substantially after this device degraded to become same or close to that of the 2nd measured new device, its channel temperature was considerably higher due to its much higher current. Same device had highest Imax before degrading and highest phonon-lifetime. It did consequently show higher channel temperature for same range of  $V_{Pulse}$  as after it degraded.



**Figure 7.** Phonon-lifetime from [6], and the self-heating assessments for the devices in "Figure 6". (a=140 × 10<sup>-13</sup>,  $b = 10^{12}$ ,  $c = 10^{10}$ )



**Figure 8.** Simulated total channel temperatures  $(T_{LT} + T_{LO,phonons})$  in HEMT-A before and after it degraded, and in the measured 2 new devices which I-V's are also shown in "Figure 6"

"Figure 9" shows the simulated currents for the devices in "Figure 6" and which simulated channel temperatures are those shown in "Figure 8". The fact that these simulations duplicated with fair accuracies the measured I-V's of these devices confirms the validity of their simulated channel temperatures that the "Figure 8" shows.



Figure 9. Simulated device currents for the devices in "Figure 6"

Low field mobility's were extracted from measured I-V's on a new device before and after we stressed it with 45V on-state DC for 30minutes. These were calculated by solving numerical derivatives of its device linear-currents using Euler's method. The equation model used in calculating these mobility's was that demonstrated in [20] and that is

$$gm = \frac{\partial I}{\partial VGS} = \mu n \times \frac{C_{AllnN}}{LG^2} \times VD + \frac{\partial \mu}{\partial VGS} \times \left( (VGS - VT) - \frac{1}{2} \times VD \right) \times VD$$
(22)

Values for  $n_{2DEG}$  were calculated from equation (21). These mobility's are shown in "Figure 10". Their observed increase with increasing  $n_{2DEG}$  is due to gradual screen-out of impurities in GaN 2DEG with higher VGS. Their decrease on the other hand at higher  $n_{2DEG}$  is caused by interface roughness scattering as the electron wave-functions shift toward the interface [20]. The observed collapse of mobility over the entire range of  $n_{2DEG}$  after device degraded indicates clearly that substantial damage incurred to the GaN 2DEG as a result. This is consistent with the substantial shifts of the VKnee's that "Figure 6" shows after devices degraded.



Figure 10. Calculated mobility's of 2DEG GaN before and after device degraded permanently from application of a 45V DC. (VD = 0.35V)

It is further observed from "Figure 10" that the onset, or the Cutoff magnitude for  $n_{2DEG}$ , at which electron scattering with the interface roughness starts predominating is same for both mobility's and is close to  $6 \times 10^{12} cm^{-2}$ . This suggests that no apparent severe damage incurred to AlN spacer, neither from the 45V DC, and that most damage is localized in the GaN. Comparative TLM measurements on same device before and after it degraded confirmed same contact resistance close to  $0.7\Omega$ .mm implying therefore that no damage incurred to metal contacts either. We believe that excess self-heating due to electron scattering with LO phonons which confined in the GaN 2DEG damaged the GaN and lowered the device current and phonon-lifetime substantially before any excessive heating was able to diffuse outside the GaN 2DEG and damage the metal contacts.

Additional DC measurements for Gate-leakage currents on same device before and after it degraded also indicated that no pronounced damage incurred to its AlInN barrier as both measured leakage currents were virtually same. If any significant damage had incurred to this barrier an increase of trap-activated tunnelling would have pronouncedly increased the Gate leakage current after device degraded. These Gate-leakage currents are shown in "Figure 11".



Figure 11. Measured device Gate leakage currents before and after device degraded permanently from application of 45V DC

We finally assessed the impacts from applying higher potentials to Drain with lower  $t_{on}$ , and hence, lower duty-cycles on the device performance. For this we defined metric  $V_{Pulse,Deg}$  as the maximum  $V_{Pulse}$  that can be applied to the Drain of device for given  $t_{on}$  and VGS before it starts degrading (e. g. for devices in "Figure 3a"  $V_{Pulse.Deg}$  are 10.5V and 20V for HEMT-A and HEMT-B respectively). Results are shown in "Figure 12". All measurements were performed on new devices that were not measured prior. Though lower values for VGS for same  $t_{on}$  are expected to increase  $V_{Pulse,Deg}$  due to the noticeable reduction of self-heating, the exact opposite is observed from "Figure 12". The  $V_{Pulse,Deg}$  did actually reduce.



**Figure 12.** Measured  $V_{Pulse,Deg}$  versus  $t_{on}$  for different values of VGS.  $(lgd = 12\mu m; t_{off} = 1s; AlInN HEMTs with 10\% In)$ 

We attribute this to the inverse-piezoelectric effect that increased the strain to the barrier and to the GaN when lower *VGS* were applied and this inverse-piezoelectric effect had apparently contributed to accelerating the crystallographic damage to an already excessively heated GaN. Though such strain is known to store mostly in the epitaxial barriers of GaN-based HEMTs, the much thinner AlInN barriers of our devices (~9nm compared to the typical 20-35nm barriers in conventional AlGaN/GaN HEMTs) can cause considerable strain to transfer to the GaN film below it.

### 4. Conclusions

We demonstrated that excess self-heating in GaN 2DEG is major contributor to the pronounced irreversible device degradation of GaN-based HEMTs that incorporate barriers having high Aluminum content. We showed that most induced damage causing this degradation appears to confine to the GaN and most pronouncedly under the access region between Gate and Drain and at the Drain side in the channel under the Gate where both electric-field and self-heating are highest. We showed that this induced damage to GaN kicks as the channel temperature approaches the known thermal limits of bulk GaN (~ 2500°K). We further showed that inverse-piezoelectric effect can further accelerate this damage to GaN when higher potentials are applied to Drain and that wider Gate-to-Drain spacing's can suppress the portion of self-heating in GaN 2DEG that is caused by electron scattering with LO phonons. We also demonstrated that by operating the devices with pulsed bias the self-heating caused by electron scattering with other than LO phonons can be effectively suppressed while the portion of self-heating due to electrons scattering with LO phonons is

more difficult to suppress as it tends to confine in the GaN 2DEG even when SiC substrates are used. Further studies are required to assess the impacts from further reducing the  $t_{on}$  on  $T_{LO,phonons}$  in devices incorporating different substrate materials; as  $t_{on}$  reduces deeper below the micro-second range higher thermal conductive substrates such as Diamond may be capable to further reduce  $T_{LO,phonons}$  even during  $t_{on}$ .

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