

Microcrack Detection in Composite Laminates at Early Stage of Thermal Cycling Using Moisture/Freeze/Dry Cycle

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Abstract Carbon-fiber reinforced polymeric composite materials have been widely used in space industry for manufacturing of spacecraft structures, satellite-panels and antennas. In space, composites are subjected to the periodic thermal cycling (TC) in which temperature ranges from -196 to 180°C , depending on operational condition. This range of temperatures causes various damages in composite, such as micro cracks, which in turn reduce the composite life cycles. Usually the micro-cracks become more evident/detectable when subjected to high number of thermal cycles. The aim of this work is to develop a technique to be able to identify the micro-crack initiation and its propagation at low number of thermal cycles. Experimental tests were conducted on composite specimens by subjecting the samples to thermal shock from -190°C (dipping in liquid nitrogen) to $+150^{\circ}\text{C}$ with different number of cycles. Then optical microscope was used to observe the micro-cracks. The microscopic results show that few micro-cracks appeared on the surface with low number of thermal cycles. Samples are then conditioned in boiled distilled water to become saturated (moisture procedure). A fast cooling, followed by sample drying in oven above water boiling temperature was performed and the samples were again examined under microscope. It was observed that the previous existing micro-cracks are more visible and also some new micro cracks surfaced. It is suggested that in order to detect the micro-cracks during the thermal cycling, moisture and heat can be used to accelerate the micro-crack growth and make them detectable even at low number of thermal cycles.

Keywords Moisture/Freeze/Dry Cycle, Thermal cycling, Composites, Micro-crack initiation

1. Introduction

Reinforced polymeric composites with high strength and low density are an alternative candidate in lightweight structures in the aerospace industry. The most important advantages of composite materials are their high strength-to-weight ratio, low density, low coefficient of thermal expansion CTE, and their inherent high thermal/temperature resistance [1]. Replacing conventional metal alloys with composites, especially in satellites, aircraft and other automotive applications, results in considerable weight savings. The objective is usually to make a component which meets the design requirements and often with a low density for longer sustainability (longer time period).

In recent years, structural parts manufactured with carbon/epoxy polymeric composite materials have been

widely used in space and aircraft industries [2-4]. In the case of satellites, temperatures can often vary by near 270°C or above as they orbit around the earth [5-8]. A satellite in eclipse region, (Earth's shadow) can experience temperatures as low as -190°C , up to 150°C in sunlit region. This phenomenon is known as thermal cycling in a cryogenic environment and is extensively found to induce micro-cracking, one of the main damage mechanisms of composite materials undergoing thermal cycling.

Thermal stresses are generated during the temperature variation and curing processes of composite materials. But in addition, moisture can induces/produce different effects such as matrix properties reductions, swelling and expansion of matrix at the fiber-matrix interface that modifies the state of residual stresses and causes the micro-crack formation in composite materials [9-11]. It is known, that the moisture has significant effects on the physical and chemical properties of epoxy matrix as well as on their final performance of composite structures especially in their long-term utilization. The absorbed water usually reduces the glass-transition temperature (T_g) by plasticizing the polymer network and also affects mechanical performance and long-term durability of high performance composites.

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Manufacturing method is an influential factor in composite material properties. Composite structures are often cured in an autoclave to achieve the required space grade quality. However, curing large structures need access to large autoclaves, which is limited and expensive. So fabrication of large structures using OOA prepreg materials will result in huge amount of savings in manufacturing costs [12]. In OOA manufacturing method, presence of voids has been an issue due to lack of high pressure onto the laminate to eliminate the voids. As an alternative, a vacuum pump is utilized to exert a low pressure for minimizing the void content [13]. Many researchers have compared OOA prepregs with autoclave materials and found that the properties of laminates fabricated with the OOA prepregs can be the same as those fabricated with autoclave prepregs [14-15]. Cauberghs *et al.* [16] studied the OOA manufacturing of aerospace representative parts with complex shape using carbon/epoxy prepreg technology and compared them with autoclave parts. The mechanical performance of OOA prepregs including compression and bending properties was comparable to autoclave prepregs. Achieving uniformity in the thickness around tight corners was reported to be challenging for both techniques.

Thermal cycling or thermal-shocking can be defined as a process of rapid changes in the temperature of the specimen from low to high or vice versa [10]. There is a lack of information in the literature and experiments need to be conducted to study the combined effects of thermal and moisture cycling loading. Space environments have a range of temperatures that cause damage in parts made of composite materials, which affect their performance and even threaten their ability to operate effectively for recommended time period. The second challenge is the orientation and lay-up of composite materials to delay the damage. Therefore, each orientation or lay-up has their own individual strength against environmental effects. The third challenge is how to analyze and simulate material behavior during test procedures, both analytically and numerically. Overall, here first priority will be given to identify the cracks and their propagation at low number of cycles, and then improve their life cycles or delay the damages with optimizing orientations/lay-up.

2. The General Concept for Microscopic Cracks Identification

The motivation for this paper is to develop a method to make micro cracks visible and detectable at early stage of thermal cycling of a polymeric composite. The concern and necessity for such a method was raised by the space/aerospace industries. Currently most of the structures in satellites or supersonic aircraft are made of composite materials or a combination of polymer components and alloys. These structures can present significant challenges for manufacturing, but most challenges are found during

their service life in fluctuating environments. Even as we consider that during manufacturing, structures contain acceptable defects or micro-cracks because of limitations in testing equipment. Micro-cracks usually become more visible and detectable at higher numbers of thermal cycling. Furthermore, the identification of micro-cracks which are considered negligible and their intensive propagation at early stage is mechanically rigorous but consequently has great potential for use in manufacturing design for complex space/aerospace applications in varying environments.

Accelerated test method is used to save time and get the approximate results to predict the damage condition for long-term behavior in short period of time. Accelerated (temperature dependent) tests are justifiable if the accelerating parameter does not cause an artificial change in the aging mechanism. To verify this assumption, tests on a reasonably long time scale are necessary. The data in [21] indicated that 1000 h of accelerated aging at 70°C caused a loss in flexural properties similar to 14-15 years in natural conditions. This is one of very few examples for which the effect of accelerated aging has been verified. Based on results of this example, proposed experimental methods of accelerated test were conducted and no damage was induced by accelerated parameters. The details procedure is described as follows.

3. Sample Preparation and Experiment

3.1. Material Selection

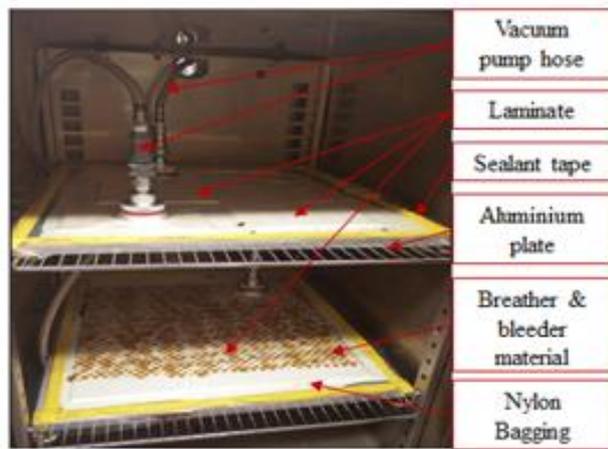
The material used for this project was Cycom 5320-1 manufactured by Cytec Engineered Materials Inc. which is a composite material system for manufacturing of primary structural parts in today's aerospace industry. It is a toughened epoxy prepreg system specifically designed for OOA manufacturing. A woven fabric of 5-harness carbon/epoxy satin prepreg (5HS) with the same resin system provided by Cytec was used to manufacture the samples, with total laminate thickness of ~3.5 mm [22].

3.2. Manufacturing of the Samples

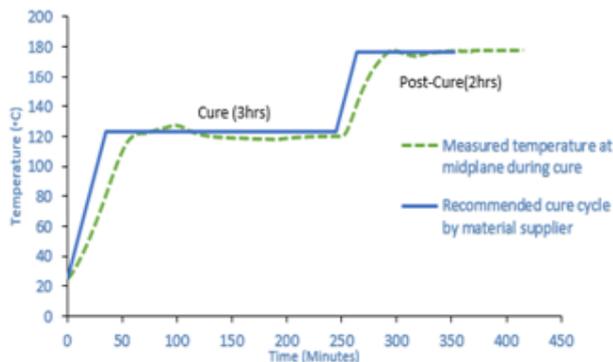
A cross-ply laminate with 8 layers of 5HS woven fabric (5-harness satin prepregs) were made. The plate was manufactured by the hand layup method and was vacuumed at a pressure of 0.1 MPa for one hour to eliminate the entrapped air and moisture. To track the plate temperature during the curing process and afterwards during the thermal cycling process, a k-type thermocouple was bonded to the edge of the mid plane of the laminate. Since the carbon fibers are conductive and by touching the thermocouple can cause false temperature reading, a release film was utilized to avoid contact between the prepreg and the thermocouple wire.

To cure the composite laminate, the cure cycle recommended by the manufacturer for Cycom 5320-1 material system was followed to cure the composite laminates [23]. The material was cured in a forced air

circulation oven. A vacuum pump was used to pull the vacuum inside the layup bag. In order to accurately measure the laminate temperature during the cure and avoid the temperature delay between the oven temperature and plate temperature, the other end of the thermocouple already embedded in the laminate mid-plane was taken out of the oven and used for data acquisition system. It was assumed that once the center ply of the laminate was within $\pm 5^\circ\text{C}$ of the desired temperature, all plies of the laminate had reached to the final temperature. Figure 1(a)-(b), shows OOA set-up and the cure cycle for the composite plate. As shown in this figure, the oven temperature was ramped from the room temperature to 120°C and held at this temperature for 3 hours to cure the epoxy. Then it was raised to 177°C and maintained for 2 hours for the post-curing, and finally the plate was allowed to cool down to room temperature. All the temperature changes occurred at a ramp rate of $0.6\text{--}2.8^\circ\text{C}/\text{min}$ (Figure 1b).



(a)



(b)

Figure 1. a) Manufacturing set-up for OOA curing inside oven, b) Measured data and recommended cure cycle by material supplier

3.3. Samples Characterization before Test

DSC test was performed on the cross-ply fabric composite plate to obtain the degree of cure. Five samples were cut from different areas of the cross-ply fabric plate in order to have an average representative of the whole plate. Figure 2

shows the DSC heat flow signal for both uncured prepreg and cured laminate as a function of temperature. A linear integration of the area under the heat flow curve was used to calculate the total heat of reaction for both uncured and cured materials, which resulted in 166.1 J/g for the raw prepreg and 3.33 J/g for the cured laminate. Based on the data obtained from DSC graphs, the average degree of cure for the cross-ply fabric laminate was about 98%, which indicate a good curing process. It should be mentioned that the degree of cure for the plate was calculated as the average value of five different samples per category.

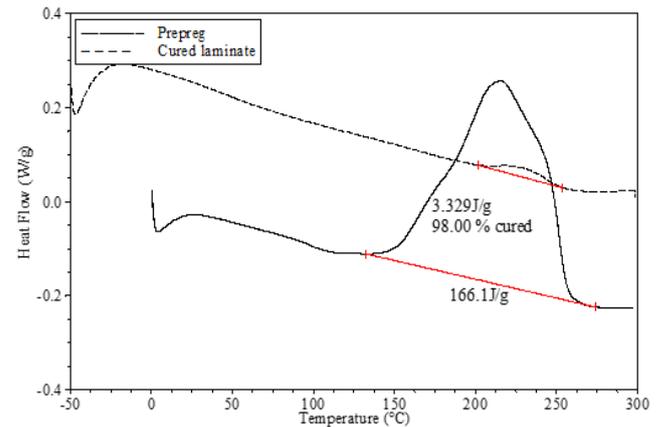


Figure 2. Heat flow of uncured prepreg and cured laminate for cross-ply fabric laminate obtained from DSC

3.4. Thermal Cycling

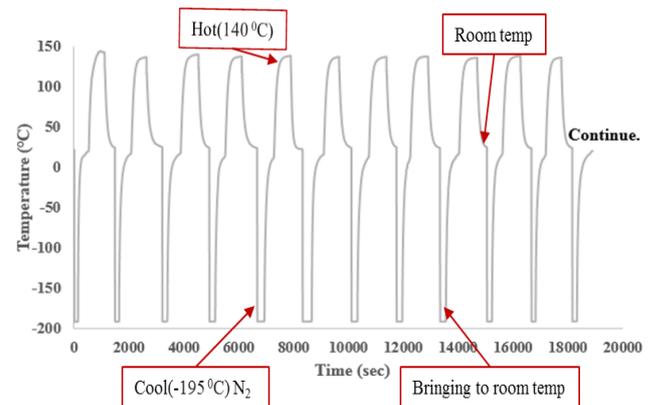


Figure 3. Thermal cycle time steps, with Temperature cycle vs. time for 10+ thermal cycles

Thermal cycling was applied on the samples with dimensions corresponding to the width, length and thickness of $30 \times 12 \times 3\text{ mm}$ respectively, using a “thermal shock” method [22]. The samples were dipped in liquid nitrogen (-195°C) for a period of 2 min and then exposed to room temperature (20°C) for a period of 5 min. Then the samples are heated up inside the oven to 150°C and maintained for 10 minutes as shown in Figure 4. This procedure will be repeated for up to 10, 50 and 100 cycles as shown in the Figure 3. Selected samples were taken under microscope after thermal cycles at 50 and 100. One side of the every sample was polished to make smooth surface for

microscopic inspection. Different cross-sections were marked for the inspection of the laminates layers, micro-cracks, voids, resin rich area and their locations inside layers or tows.

3.5. Moisture/Freeze/Dry Cycle

As of 100th thermal cycle done, an accelerated moisture/Freeze/Dry Cycle was designed and developed. It was assumed that the accelerated moisture test cause a very similar moisture response in the material as the baseline environment, but in a much shorter time. Similarly, the accelerated thermal cycle is predicted to cause thermal micro-cracking damage equivalent to that caused by the baseline environment in a much lower number of more intense thermal cycles. After applying a number of thermal cycles, the specimens are immersed in distilled water in a glass vessel. Then the glass vessel was immersed in a glass container which holds heated oil with the desired temperature which is controlled by heater from the Heidolph MR Hei-Standard Company (Figure 4). In order to measure the amount of water absorbed within each samples, the specimen weight has to be measured periodically by an analytical balance (Sartorius Company) which is accurate to ~0.1 mg. Before weighing, the specimens had been wiped out with a paper towel and dried for approximately 1-2 min to remove any excess surface moisture and to avoid any error. The weight and conditioning time should be recorded to calculate the moisture content.

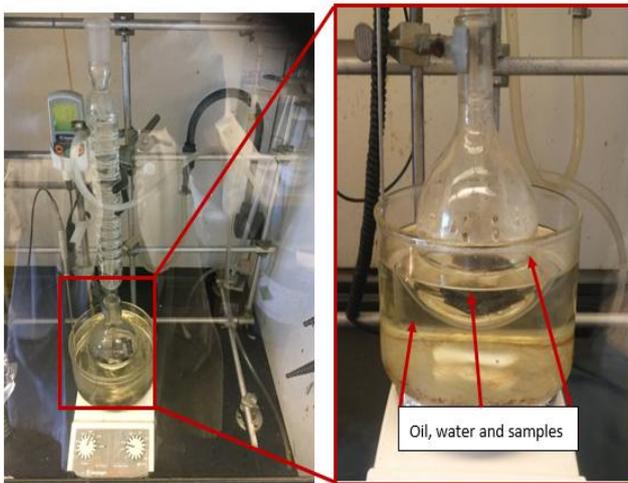


Figure 4. Moisture set up

The weight gain of the test samples after exposure to the selected relative humidity environments is measured as a function of time. The humidity content is estimated using the following equation.

$$M = \frac{M_w - M_d}{M_d} \times 100 \quad (1)$$

Where 'M' is the percentage of gained humidity, M_w is the weight of the wet sample, and M_d is the weight of the dry sample. The weight of the dry samples, i.e., the weight of the

samples in the room temperature with relative humidity chamber before moisture in this case. At the beginning of the conditioning period, it is observed that the samples had constant weight. When the samples did not change any weight, then they are transferred to the relative humid chambers and their weight gain is continuously monitored.

Composites samples are immersed in the distilled boiling water. The weight gains are calculated and plotted as a function of square root of time. Each point represents an average from the measurement of three specimens as shown in Figure 5. Cross-ply appears to be saturated with 1.2% within the time frame considered. Specimen weight has been measured every 6 hours until saturation happened or until more than 1% of moisture weight has been gained as it shown in graph in Figure 5.

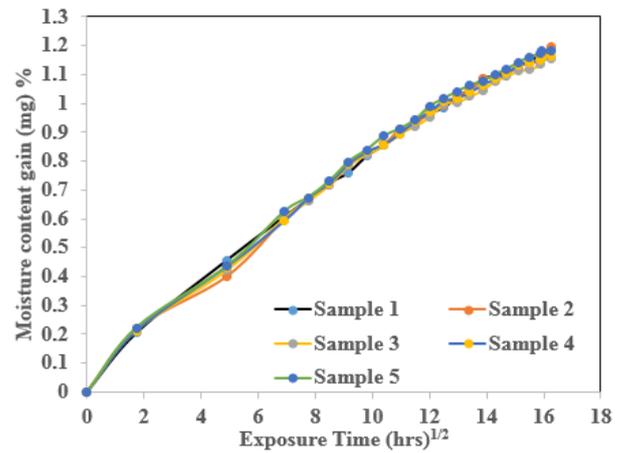


Figure 5. Weight Gain of the samples as a Function of time

3.6. Freeze/Dry Processing

Freezing and drying processes are used to open more cracks as saturated samples are filled with water that expand over water freezing and boiling point temperatures. After moisturization cycle, specimens are kept in liquid nitrogen for freezing them and again weight was measured. After freezing, the specimens are analyzed under microscope and then they were taken into the oven at 110°C until their weight loss has stabilized and no further weight loss of the sample could be observed. This condition of the specimens is referred to as dry.

Onset cracks developed by thermal cycle are saturated faster (during saturation processes) that help to accelerate crack propagation. These saturated samples are immersed in liquid nitrogen to freeze the saturated water in their own cavity or over void surfaces. Initial cracks are propagated faster during drying processes as saturated liquid expand their volume under increasing temperatures. Again, samples are needed for the purpose of crack investigation using optical microscope. Those samples are analyzed after moisture cycle to check the new crack initiation and propagation during wet cycles, and afterwards they are analyzed for freezing and dried cycles.

4. Results and Discussion

The cross-sectional regions in each sample were examined under microscopic with 10-20X- magnification to track the cracks/damage in the entire sample and to detect the presence of micro-cracks. After this, every single layer were observed at specific locations for analysis of the path of the crack. During image analysis, we were also focused on crack onsets, which appear in the form of voids, which would result in cracks/delamination on the inner and outer ply of the composite laminate. Figure 6(a) shows selection of cross-sectional area, here no micro-cracks are visible. There are some void that could be the cause for the cracks initiation under thermal cycles. It can be observed that after 100th thermal cycles, the cracks become slightly more visible (Figure 6(b)). It does not mean that the micro-cracks appeared suddenly near 100th cycles. It could be there but as residual stresses increased by “thermal shock” like cooling and heating, aided to open the shield surface as a micro-crack.

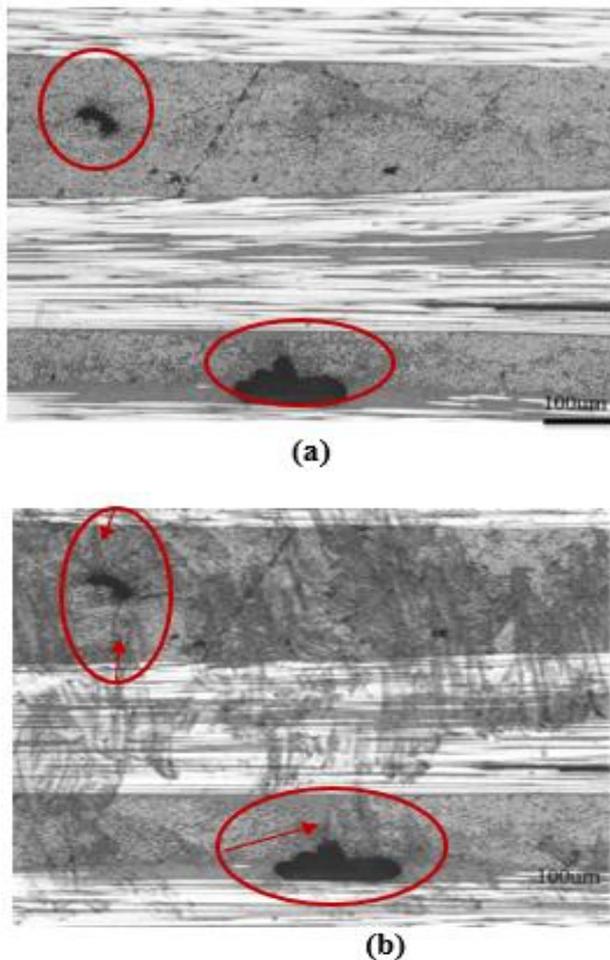


Figure 6. Micrograph and crack detection a) before thermal shocking, b) after 100 thermal cycles

An accelerated moisture cycle was designed as discussed, which is expected to cause a very similar micro-crack response in the material as the baseline environment but in a

much shorter time. All the water, which was absorbed by matrix, inside cavity, void, interface laminate layers will increase the density or expand the samples. The measurement by the weight shows that the all samples were saturated, then we took next step to freeze the samples by immersing in the liquid nitrogen (-195°C). Time period for freezing was 2-3 min and bring it back to room temperature. After freezing, the microcracks in the samples were analyzed under microscope to investigate again the crack propagation. Through the image analysis we observed onset cracks, which were more prominent as shown in Figure 7(a).

After the examination of the frozen samples under microscope, samples were dried by heating them above 100°C in oven. Dried sample were again examined under microscope and it was observed that the cracks grew and become larger. This process can force the cracks to open more and therefore more invisible cracks might become visible as shown in Figure 7(b).

The crack tips developed by thermal cycling were saturated faster (during saturation processes) that help to accelerate crack propagation. These saturated samples are immersed in liquid nitrogen to freeze the saturated water in their own cavity or over void surfaces. Initial cracks are propagated faster during drying processes as saturated liquid expand their volume under increasing temperatures.

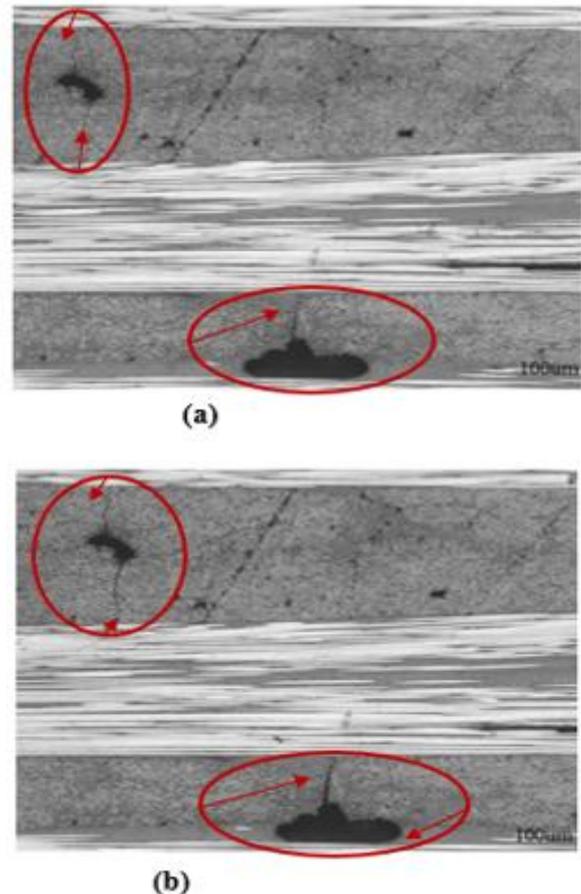


Figure 7. Crack detection a) after moisture saturation and freezing and b) after drying

Figure 8 shows another sample with void on the resin rich area between the layers. No crack is observed at the beginning. After freezing the sample in liquid nitrogen, the micro-cracks become more visible (Figure 8(c)). After drying the sample in the oven, cracks become more visible and their width become larger.

It is difficult to predict theoretically the effect of voids on the physical strength of composite laminates because of their irregular size, shape and location. Therefore, predicting the

voids effect for the prediction of full structure behavior of the composite laminates is necessary and was analyzed here. To investigate the influence of voids, laminates with void content about 0.5% have been examined under microscope for different number of thermal cycles. As an example, Figure 9 shows the laminate cross-section after 100 thermal cycles. These images are used to count the number of microcracks at void free area and also around the voids.

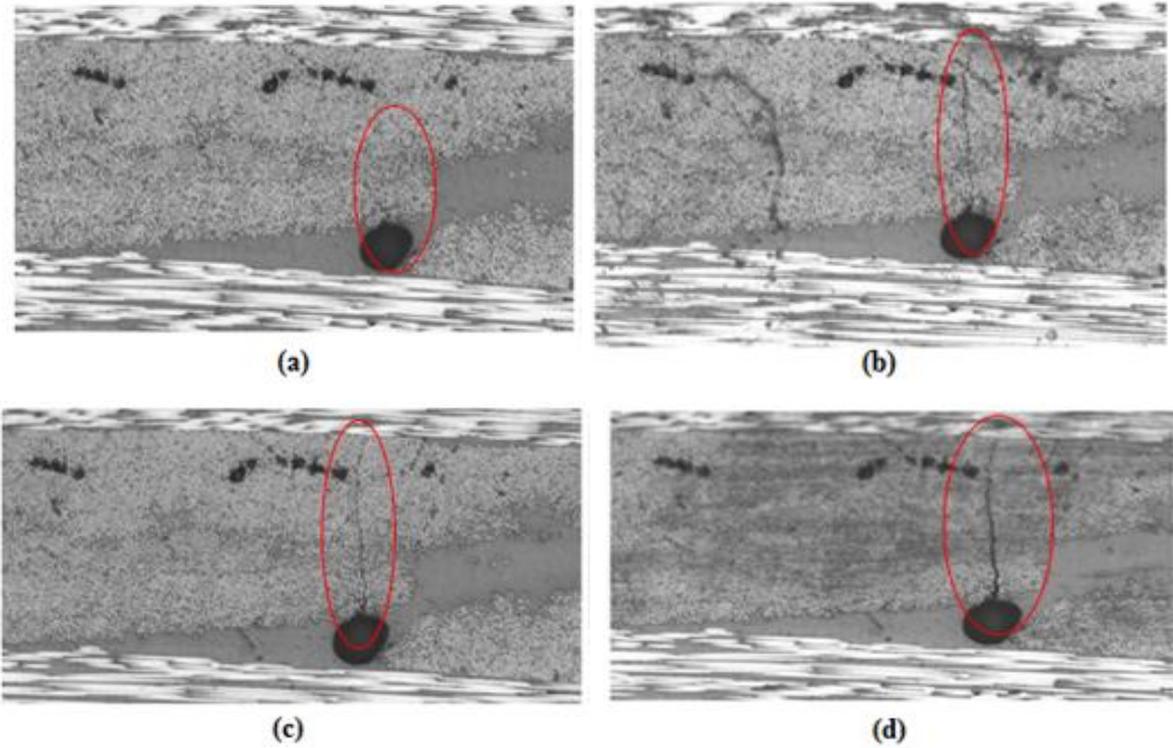


Figure 8. Micro-cracks formation from the void surface (a) before TC (b) after 100 TC (c) N₂ treatment and (d) Dry in oven

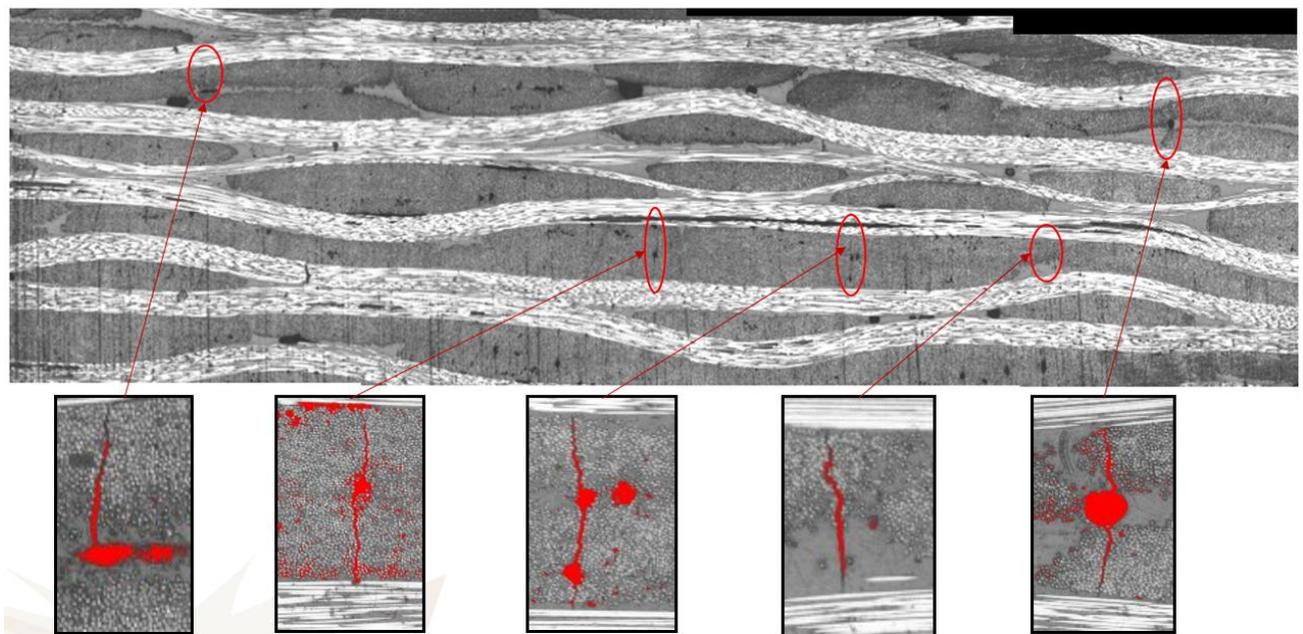


Figure 9. Effect of Void on Micro-Cracks Formation and Propagations

Figure 10 shows the number of visible microcracks for the samples that are subjected to different thermal cycles and then followed by moisturizing, freezing, and drying process. The visual inspection of the same cross-section under microscope is the method that has been used to identify the new cracks from the older ones. For the moisture/freeze/dry cycle effect, first 100 numbers of thermal cycles were performed on the sample and then one moisture/freeze/dry cycle is performed. The number of visible microcracks was counted. As can be seen, the number of microcracks increases by increasing the number of cycles. In addition, it is noted that the number of microcracks around the voids are more compare to the void free areas which is an indication of void effect on the microcrack formation. After applying one moisture/freeze/dry cycle, the number of microcracks jumps to the total of 93, almost three times of the microcracks at the beginning. It shows how this cycle can accelerate the crack propagation. It seems that the penetration of the water into the polymer chain, which reduces the mechanical properties, can be one of the reasons for this behavior. At the same time, penetration and presence of water inside the void and inside the microcrack space followed by freezing and drying (above 100°C), could be other reasons, which force the microcracks to move faster. Water volume increases during the freezing as well as boiling and evaporation. This increase in volume can generate internal pressure, which can cause the microcracks move faster. More work has to be done to identify the real mechanisms behind this behaviour.

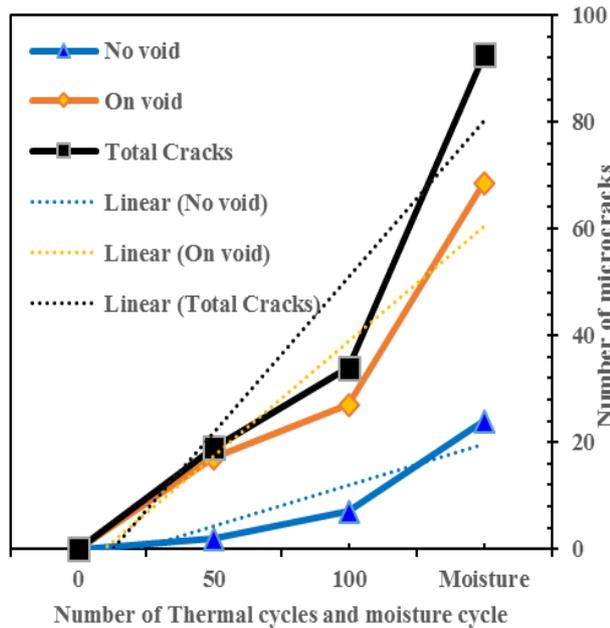


Figure 10. Number of cracks with TC and moisture

4.1. Interlaminar Shear Strength ILSS Results

To investigate the effect of microcracks on the material properties, ILSS test are conducted after different number of thermal cycles as well as after the moisture cycle to compare their strength. Figure 11 shows the variation of the ILSS of original carbon/epoxy composites versus the samples that are

conditioned under different thermal cycles and also with the sample which went through moisture/freeze/dry cycle. The ILSS of the cross-ply fabric samples show that as the number of thermal cycles increases from 0 to 100, ILSS of the composites gradually decreased from 63.39 MPa for the unexposed sample to 59.57 MPa for the 100-times-cycled sample. ILSS of this sample (100-cycled sample) after exposure to the moisture/freeze/dry cycle reduces to 56.48 MPa.

This observation is consistency with the previous experimental results related to the number of microcracks formation. Application of moisture/freeze/dry cycle increases the number, length and width of the microcracks, which in turn reduce the composite mechanical properties.

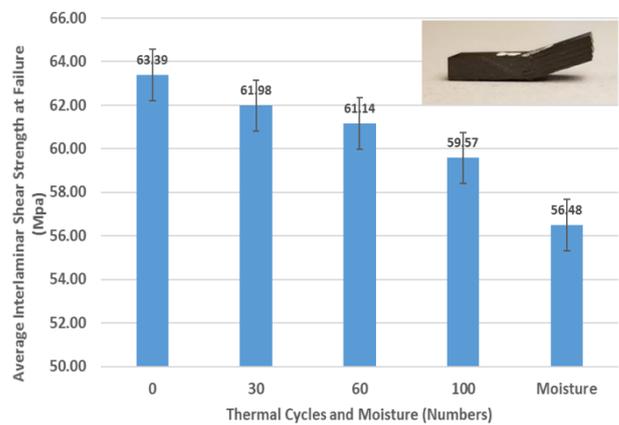


Figure 11. Interlaminar shear strength with respect to thermal cycle and moisture cycle

Figure 12 shows the variation of number of microcracks and ILSS versus the number of thermal cycles. Also the results related to the moisture/freeze/dry cycle sample are added for the comparison. It can be seen that the reduction of ILSS with respect to number of cycles within the test range is linear. However, it should be mentioned that the development of number of crack and the crack length with number of cycles is similar to the thermal cycle samples.

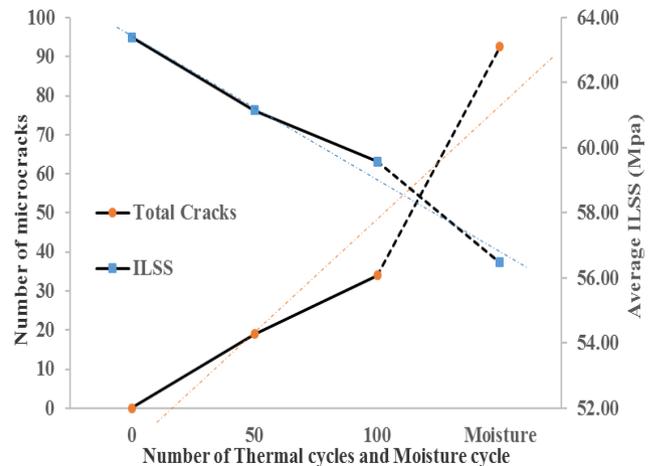


Figure 12. Microcracks numbers and strength variation with thermal and moisture cycles

4.2. Elastic Properties after Thermal Cycle and Moisture/Freeze/dry Cycle

Dynamic Mechanical Analyzer DMA has been used to study the effect of thermal cycle and moisture/freeze/dry cycle on the elastic and damping properties of composite. Figure 13 shows the variation of storage modulus E' of the cross-ply composite with respect to temperature for samples subjected to the different number of cycles. The storage modulus at room temperature tends to significantly decrease from 0 to 100 thermal cycles, showing that the majority of damage and microcrack formation and propagation happen within this range. There is not much difference between the elastic modulus of the 100-cycled samples with and without moisture/freeze/dry cycle. One reason could be that the moisture/freeze/dry cycle may cause propagation of the existing microcracks on the surface (and therefore makes them more visible) but apparently has no contribution in creation of new microcracks, particularly inside the laminate.

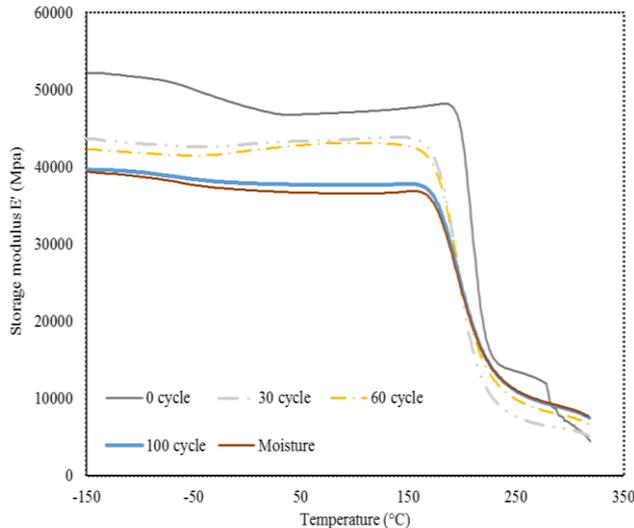


Figure 13. Effect of thermal cycling and moisture/freeze/dry cycle on storage modulus of cross-ply fabric laminate

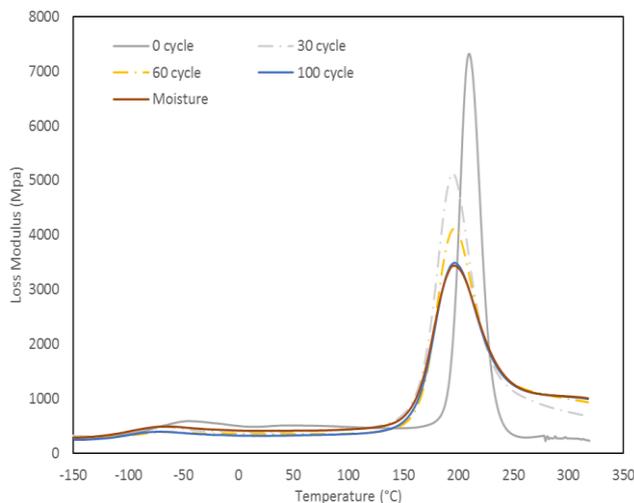


Figure 14. Effect of thermal cycling and moisture cycle on the loss modulus of cross-ply fabric laminate

Figure 14 exhibits the effect of the thermal cycling and moisture/freeze/dry cycle on the loss modulus (E'') of different samples. Though looking at the peak of the loss modulus, it can be observed that it fell significantly by increasing the number of cycles. The reduction rate is very significant at early cycles, which is near 30% less than $E''_{\text{unexposed}}$. This reduction rate slows down by further cycling up to 100 cycles, which is about 52% less than unexposed. It is clear from the figure that with increasing the number of thermal cycles, the magnitude of the peaks decreases while the curve at peak area gets broader. In addition, it can be observed that applying the moisture/freeze/dry cycle does not have a significant effect on the damping properties.

5. Conclusions

Identification of microcracks at early stage of thermal cycling is difficult due to their small sizes. In this study, an experimental method called moisture/freeze/dry cycle is developed in which, the samples after thermal cycling are dipped into boiled distilled water to introduce moisture into matrix and voids and cracks. After which the sample is frozen in the liquid nitrogen for about 5 min, then heated up to 110°C in order to remove the moisture content. Introduction of the moisture to the samples after the thermal cycling changes the matrix properties. Moisture can penetrate inside the void and cracks. The induced frozen step of the developed method helps in the transformation of absorbed water inside the voids to the solid form (ice). When the samples are dried, the ice formed during the previous step tends to evaporate. Freezing and evaporation of water inside the void and cracks jointly with reduction of the matrix stiffness due to the moisture absorption helps the crack moves and propagates. It was observed that the previous existing micro-cracks are more visible and also some new micro cracks surfaced. It is demonstrated that, to detect the micro-cracks at low number of thermal cycles, moisture and heat can be used to accelerate the micro-crack growth and make them more detectable. Thus, the damages can be detected at early stage of its formation, which saves time.

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REFERENCES

- [1] W. D. Callister and D. G. Rethwisch, Materials science and engineering: An introduction. 2009.

- [2] S. Rouquie, M. C. Lafarie-Frenot, J. Cinquin, and a. M. Colombaro, "Thermal cycling of carbon/epoxy laminates in neutral and oxidative environments," *Compos. Sci. Technol.*, vol. 65, no. 3–4, pp. 403–409, 2005.
- [3] R. A. Schapery, "Thermal Expansion Coefficients of Composite Materials Based on Energy Principles," *Composites*, vol. 15, no. 2, p. 161, 1984.
- [4] G. Yu, D. Shang-Li, H. Song, Y. De-Zhuang, and L. Zhi-Jun, "Characterization of Stress Distribution and Thermal Expansion Behavior for M40J/AG-80 Composites Experienced Vacuum Thermo-cycling," *J. Reinf. Plast. Compos.*, vol. 25, no. 16, pp. 1647–1657, 2006.
- [5] F. E. Penado, "Effective Elastic Properties of Honeycomb Core with Fiber-Reinforced Composite Cells," *Eng. Fract. Mech.*, vol. 2013, no. October, pp. 89–96, 2013.
- [6] S. EZhandarov and E. Episanova, "Investigation of Load Transfer Between the Fiber and the Matrix," *Eng. Fract. Mech.*, no. S. EZhandarov, E. Episanova, E. Emsder, and J. EA. ENairn, 2005.
- [7] B. Pal, "Analytical Estimation of Elastic Properties of Polypropylene Fiber Matrix Composite by Finite Element Analysis," *Adv. Mater. Phys. Chem.*, vol. 02, no. 01, pp. 23–30, 2012.
- [8] S. Bhaskara, R. Devireddy, and S. Biswas, "Effect of Fiber Geometry and Representative Volume Element on Elastic and Thermal Properties of Unidirectional Fiber-Reinforced Composites," *Eng. Fract. Mech.*, vol. 2014, 2014.
- [9] J. I. Cauich-Cupul, E. Pérez-Pacheco, A. Valadez-González, and P. J. Herrera-Franco, "Effect of moisture absorption on the micromechanical behavior of carbon fiber/epoxy matrix composites," *J. Mater. Sci.*, vol. 46, no. 20, pp. 6664–6672, 2011.
- [10] S. Zhu et al., "Effect of moisture on the measured tensile strength of polyacrylonitrile carbon fibers," *J. Mater. Sci.*, vol. 51, no. 5, pp. 2371–2379, 2016.
- [11] D. J. Boll, W. D. Bascom, and B. Motiee, "Moisture absorption by structural epoxy-matrix carbon-fiber composites," *Compos. Sci. Technol.*, vol. 24, no. 4, pp. 253–273, 1985.
- [12] R. W. S. and D. A. D. A. Shanker, "Elastic properties of CYCOM 5320-1/T650 at elevated temperatures using response surface methodology," *Compos. Hybrid, Multifunct. Mater. Springer*, vol. 4, pp. 29–37, 2015.
- [13] K. Jackson and M. Crabtree, "Autoclave quality composites tooling for composite from vacuum bag only processing," 47 Th Int. SAMPE Symp. Exhib. 2002, pp. 800–807, 2001.
- [14] B. W. Grimsley et al, "Elevated temperature, notched compression performance of out of autoclave processed composites," NASA Langley Res. Center, Hampton, VA, United States, 2013.
- [15] J. K. Sutter et al, "Comparison of autoclave and out-of-autoclave composites," NASA Res. Cent., 2010.
- [16] Cauberghs and Julien, "Out-of-Autoclave Manufacturing of Aerospace Representative Parts," 2012.
- [17] W. S. J. Tompkins SS, "Effects of thermal cycling on mechanical properties of Graphit Polimide," *Mater. Lett.*, vol. 58, no. 12–13, pp. 1899–1902, 2004.
- [18] J. A. Sanborn and D. E. Morel, "Effects of Thermal Cycling on the Mechanical and Physical Properties of a Space Qualified Epoxy Adhesive," *J. Reinf. Plast. Compos.*, vol. 7, no. 2, pp. 155–164, 1988.
- [19] Q. Yu et al., "Effects of vacuum thermal cycling on mechanical and physical properties of high performance carbon/bismaleimide composite," *Mater. Chem. Phys.*, vol. 130, no. 3, pp. 1046–1053, 2011.
- [20] I. Tsukrov, B. Drach, and T. S. Gross, "Effective stiffness and thermal expansion coefficients of unidirectional composites with fibers surrounded by cylindrically orthotropic matrix layers," *Int. J. Eng. Sci.*, vol. 58, pp. 129–143, 2012.
- [21] R. Choqueuse, P.D., Davies, F., Baizeau, "Aging of Composites in Water: Comparison of Five Materials in Terms of Absorption Kinetics and Evolution of Mechanical Properties, High Temperature and Environmental Effects on Polymeric Composites," *ASTM STP 1302*, vol. 2nd Volume, pp. 73–97, 1997.
- [22] S. K. Gupta and M. Hojjati, "Thermal cycle effects on laminated composite plates containing voids," *J. Compos. Mater.*, p. 002199831878678, 2018.
- [23] Cytec, "cycom® 5320-1 epoxy resin system technical data sheet," 2015.