

# Characterisation and the Influence of Voids on Flexural and Tensile Modulus of Layered Silicate/Vinyl Ester Nanocomposites

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**Abstract** In this work, the flexural and tensile moduli of the neat vinyl ester and the corresponding nanocomposites are investigated. In addition, the influence of the voids content of the nanocomposites on the mechanical modulus is presented. Various amounts of clay were used in this work which include 2, 4, and 5 wt.% clay loading. To characterise interlaminar structure of the nanocomposites, XRD, SEM and TEM were performed. The mechanical modulus of the neat sample was improved by the incorporation of layered silicate up to 4 wt.%. Further addition of layered silicate resulted in decreasing the modulus properties. This was due to the presence of high numbers of microvoids in the nanocomposites structure at high clay loading. The increment in microvoids content was proportional to the clay loading. At 2 wt.% clay content, the voids were about 2.45 %. However, at 4 and 5 wt.%, the voids amount represented 6.40% and 9.03 % respectively.

**Keywords** Nanocomposites, Vinyl ester, Layered silicate, Microvoids

## 1. Introduction

Polymers with various particulate fillers have been successfully reinforced to improve their mechanical, thermal, and barrier properties [1-5]. The presence of the traditional fillers usually results in unwanted properties such as brittleness and opacity. Thus, polymer-clay nanocomposites (PCN) have attracted considerable interest because of the enhancements in various properties without sacrificing the opacity and brittleness [6]. PCN often provide more attractive improvements to material properties than both micro and macro composite materials [7-12]. The improvements include mechanical, thermal, barrier, and flame retardance properties. In addition, the incorporation of layered silicate into the polymer matrix usually results in improvement of the polymer performance [13]. These improvements can be achieved only by the addition of less than 5 wt.% clay into the neat matrix [14]. The type of clay used, the kind of pre-treatment, the polymer component chosen and the way in which the nanocomposite incorporates the polymer, can have a profound influence both on the properties and the structure of the nanocomposite [15, 16].

The reason for improvement of the material properties is the aspect ratio of the layered silicate where the interfacial

interaction between the organically modified layered silicate and the matrix is higher than traditional composites. Layered silicate has a thickness of sheet in the order of 1 nm and a very high aspect ratio. A little weight percent of OMLS is dispersed throughout the matrix, thus giving a larger surface area for polymer filler interfacial interaction than in conventional composites [16-18].

In this context, the investigation of the mechanical properties, including tensile and flexural modulus, of the neat vinyl ester and corresponding nanocomposites took place. Moreover, this study aims at analysing the influences of microvoids existing in the nanocomposites structure in the flexural and tensile modulus.

## 2. Experimental

### 2.1. Materials

The matrix material used in this study is vinyl ester (VE) resin. "Vinyl ester resins are oligomers resulting from the reaction between bisphenol-A based epoxy oligomers and unsaturated carboxylic acids, such as acrylic or methacrylic acid, which provide unsaturated terminal sites" [19]. This material was purchased locally and commercially coded as AME 6000 T 35. The layered silicate that has been used is Cloisite® 10A which is classified as a natural montmorillonite that is modified with a quaternary ammonium salt; it was purchased from Southern Clay Ltd. This clay can be used to improve different physical

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properties such as barrier, flame retardance and reinforcement[20].

## 2.2. Sample Fabrication Process

### 2.2.1. Neat Vinyl Ester

In order to make neat vinyl ester panels, the vinyl ester was directly mixed with the curing agent (Methyl ethyl ketone peroxide(MEKP) (mix ratio 1.5%) and then was poured in a steel mould. The mould was closed and the composite panel was left to cure in a hydraulic press at a temperature of 55°C and at a compaction pressure of 1 MPa for 2 hours.

### 2.2.2. Nanocomposites

Combinations of melt intercalation method with the compression moulding method were used to fabricate nanocomposite panels. The vinyl ester resin was mixed with various concentrations of nanoclay at room temperature using a mechanical mixer in an ultrasonic bath for 2 hours. A degassing process was applied in the mixture for 3-4 hours then it was left overnight in order to get rid of the remaining air bubbles naturally. A curing agent (MEKP) was added to the mixture (1.5%) with further gentle mixing before transfer of the mixture to the steel mould. The mould was closed and the composite panel was left to cure in a hydraulic press at a temperature of 55°C and at a compaction pressure of 1 MPa for 2 hours. A post curing process of neat and nanocomposites samples were followed at 80°C for 3 hours. The clay loadings were 0, 2, 4, and 5 wt.%.

## 2.3. The Calculation of Nanocomposites Voids Volume

Nanocomposites voids were calculated by using the following equation;

$$V_V = 1 - \rho_n \left( \frac{w_c}{\rho_c} + \frac{w_m}{\rho_m} \right) \quad (1)$$

Where  $V_V$  is the volume fraction of voids,  $\rho_n$  the density of nanocomposites,  $w_c$  the weight percent of clay (%),  $\rho_c$  the density of clay g/cm<sup>3</sup>,  $w_m$  the weight percent of matrix (%) and  $\rho_m$  is the density of matrix g/cm<sup>3</sup>. The density of nanocomposites was calculated as following;

$$\rho_n = \rho_c V_c + \rho_m V_m \quad (2)$$

where  $V_c$  and  $V_m$  are the volume fractions of clay and matrix, respectively

## 2.4. Characterisation

### 2.4.1. Wide Angle X-ray Diffraction (WAXD)

WAXD analysis on compression-moulded specimens was used to determine the clay intercalation and interlayer spacing utilising a Philips APD 1700 X-ray diffraction system with Cu K $\alpha$  radiation ( $\lambda = 1.542\text{\AA}$ ) generated at 40mA and 40 kV. The basal-spacings (the d-spacing, in Angstroms, between layers) were calculated using Bragg's Law.

### 2.4.2. Scanning Electron Microscopy (SEM)

The morphology of vinyl ester /nanocomposite systems was investigated in a Hitachi S4500 SEM working at an operating voltage of 8 kV. Block faces were prepared from each material then ultrathin sections (63nm) were collected using a diamond knife in a Reichert Ultracut E ultramicrotome. Plasma etching was used to preferentially remove the vinyl ester matrix and leave the clay particles sitting proud of the surface. After adhering to SEM stubs, a thin layer of gold/palladium was applied to the specimens prior to examination in a Quanta 250 FEG SEM.

### 2.4.3. Transmission Electron Microscopy (TEM)

TEM measurements on vinyl ester/nanocomposite systems were performed using a high-resolution transmission electron microscope (Phillips CM12 with an associated Gatan digital camera system). The same block faces used to produce the sections for SEM examination were also used for TEM.

## 2.5. Testing

### 2.5.1. Flexural Test

The flexural properties modulus of the neat vinyl ester and the corresponding nanocomposites samples were investigated by using the 3-points bending test process under the specification of BS EN 2747:1998[21]. A span of 48 mm length was utilised in a 30 kN load cell. The load was applied midway between the supports. The speed of crosshead was 2 mm/minute. The neat and nanocomposites samples were loaded until any failure was observed and then their average values were calculated.

### 2.5.2. Tensile Test

The tensile modulus of neat polymer and nanocomposites samples were carried out at a crosshead speed of 10 mm/minute under BS EN 2747:1998[22]. The tensile test specimens were prepared by utilising a water jet into rectangle beams from the nanocomposites slabs fabricated by a compression moulding method.

## 3. Results and Discussion

### 3.1. Nanocomposites Voids Content

The voids content in nanocomposites is a big concern for the industrial or engineering designs. The pre-failure and the promoting of the local deformation of the applications can be obtained by the existence of high content of microvoids[23]. Thus the study of the parameters which influence the content voids as well as the percentage of these values can help to improve or maintain the quality of samples fabricated. The voids content in the nanocomposites sample were calculated using equations (1) and (2) and the results are presented in Table 1. The amount of voids was increased with the increase in clay loading.

**Table 1.** Voids content of different nanocomposites samples

Sample	Void content (%)	Increment (%)
Vinyl ester + 2.00 wt.% clay	2.45	0
Vinyl ester + 4.00 wt.% clay	6.40	1.61
Vinyl ester + 5.00 wt.% clay	9.03	2.68

### 3.2. Characterisations of the Interlamellar Structure and Surface Morphology

#### 3.2.1. Wide Angle X-ray Diffraction (WAXD)

Wide Angle X-ray Diffraction (WAXD) is a method broadly utilised in the study of intercalation or exfoliation which additionally characterises nanocomposites and the study of the interaction between layered silicate and the matrix. Nanocomposites show dramatic enhancement in properties when neat polymers are compared with polymers based on modified layered silicate. These improvements are attributed to the sufficient dispersal of layered silicate within a polymer matrix. A micro-composite structure is obtained when a weak interaction occurs between the clay and the matrix. X-ray diffraction is utilised to show the intercalation or exfoliation structures by calculating the intergallery spacing, in order to identify the structure of the nanocomposite[24]. In this section, the amount of d-spacing of the intergallery is presented and discussed for different clay loading nanocomposites as well as for neat polymer.

**Table 2.** XRD results obtained from different clay loading of nanocomposites

Sample No.	2 $\theta$ values at 20°	The interlayer distances (nm)	d-spacing improvement %
Cloisite 10 A	20.00	0.443	00.00
VE + 2.00 wt.% clay	16.86	0.525	18.51
VE + 4.00 wt.% clay	13.84	0.639	44.24
VE + 5.00 wt.% clay	16.08	0.551	24.38

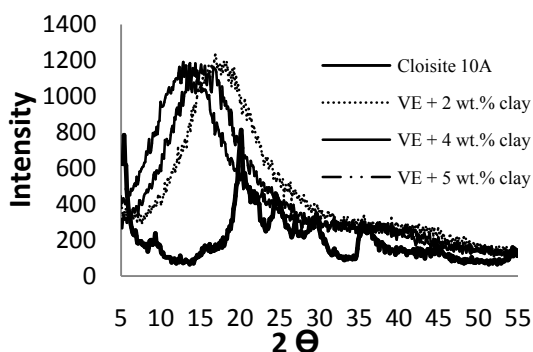
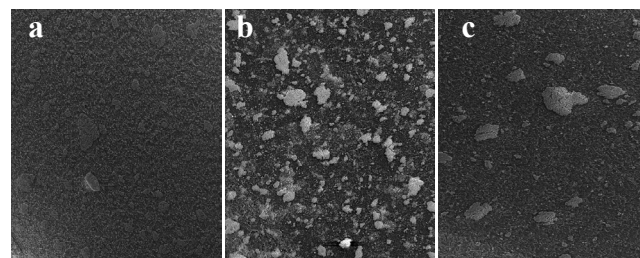
**Figure 1.** XRD results of neat polymer and the corresponding nanocomposites

Table 2 represents the change of the basal spacing ( $d_{001}$  spacing) of different samples of nanocomposites that were calculated by Bragg's Law using the values extracted from the XRD curves. From Table 2 and Figure 1, it can be seen that the nanoparticles reinforced samples show various x-ray diffraction patterns. The  $2\theta$  value for only Cloisite 10 A was  $20^\circ$  which represents 0.443 nm basal distance. The first

peak at  $2\theta$  value of  $16.86^\circ$  (2% w/w clay reinforced sample) illustrates the partial intercalated d-spacing of the clay at approximately 0.525 nm with an improvement of the d-spacing about 18.51% compared to base clay. By the addition of 4 wt.% layered silicate, the  $2\theta$  exhibited less amount than the previous clay loading which was  $13.84^\circ$  and displayed 0.639 nm of the interlayer spacing. The peak for the 5 wt.% clay loading sample at  $2\theta$  value has shifted towards a higher angle ( $16.08^\circ$ ) which indicated less intercalated d-spacing of 0.551 nm.

A clear relationship between the gallery distance and the level of dispersion of the clay in the matrix is proved by the  $2\theta$  values. In addition, the higher amount of interlayer distance, the less intercalated structure. Thus, the improvement in basal spacing led to enhancing the overall properties as will be discussed later. In this study the 4wt.% clay loading exhibited the highest value of  $d_{001}$  spacing.

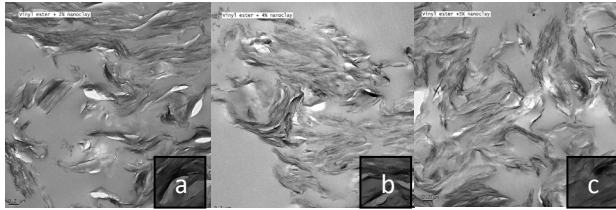
#### 3.2.2. Scanning Electron Microscopy (SEM)

**Figure 2.** SEM images at 300  $\mu\text{m}$  of (a) 2 wt.%, (b) 4 wt.% and (c) 5 wt.% nanocomposites

The SEM examination in Figure 2 shows clearly the distribution of the clay through the vinyl ester for each of the three levels of loading. As the selected images show below, the largest clay agglomerates are of a similar size for all three samples, being around 30 to 35 microns in size. However, their frequency increases with increase in loading, as does the degree of infilling between them with smaller agglomerates. Also, the voids were found to be increased with the increase in the clay content. It can be seen that the 2 wt.% clay loading shows non pronounced stacked layers and uniform distribution. The addition of further clay (i.e 4 wt.%) showed uniform distribution of particle with little sheets agglomeration. The SEM image of 5 wt.% clay loading exhibited a high number of stacked clay particles and voids compared to 2wt.% and 4 wt.% clay.

#### 3.2.3. Transmission Electron Microscopy (TEM)

Figure 3 shows the TEM micrographs of 2, 4 and 5 wt.% nanocomposites samples at 0.2  $\mu\text{m}$  magnification, where the bright region represents the matrix sea and the dark lines correspond to the stacked or individual silicate layers. Indications are from the higher magnification images that greater levels of exfoliation of the clay particles are achieved with lower nanoclay loading. At 2 wt.% clay loading, the TEM image indicates good dispersion of layered silicate throughout the polymer matrix.



**Figure 3.** TEM micrographs at 0.2μm magnification of (a) 2 wt.%, (b) 4 wt.% and (c) 5 wt.% nanocomposites

Exfoliated/ intercalated structure was obtained by the presence of 4 wt.%. At high amount of clay (i.e. 5 wt.%), additional dark areas are observed indicating the stack silicate layers and insufficient uniform dispersion. This was attributed to the high viscosity of the mixture where the ability of dispersing the layered silicate and the polymer is restricted. It is acceptable that the higher amount of clay loading mixed with the polymer, the less exfoliated and aggregated nanocomposites structure[25-27]. These findings support the results by SEM.

### 3.3. Flexural Modulus

**Table 3.** Flexural test results

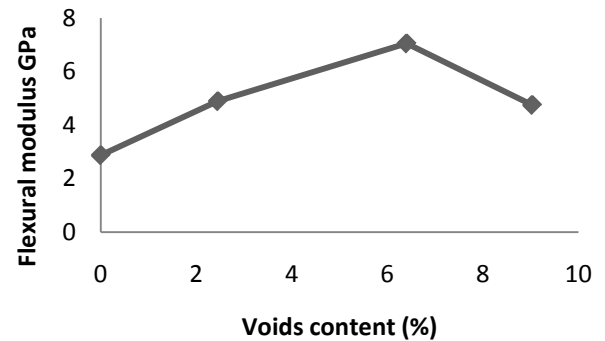
Samples	Flexural Modulus GPa SD %	Improvement in flexural modulus %
Neat Vinyl ester	2.87+/- 9	0.00
Vinyl ester + 2.00 wt.% clay	4.89+/- 5	70.15
Vinyl ester + 4.00 wt.% clay	7.05+/- 4	145.35
Vinyl ester + 5.00 wt.% clay	4.75+/- 8	65.21

The average results obtained from the four specimens tested are represented in Table 3. These show a significant enhancement in flexural modulus for nanocomposites compared to neat matrix samples. The flexural modulus for the neat sample were 2.87GPa, whereas for 2 wt.% clay loading the modulus was 4.89GPa, which was an improvement of about 70%. At 4 wt.% clay loading, the flexural modulus represented 7.05GPa. The addition of more clay (i.e 5 wt.%) showed a reduction in the modulus and exhibited 4.75GPa. The significant improvement in the 4wt.% clay loading is attributed to the properly dispersed layered silicate within the host polymer as proved by XRD, SEM, and TEM.

As seen in the Table, the layered silicate had a strong effect on the resulting properties where the enhancement in flexural modulus were proportional to the clay content up to 4 wt.% clay loading. The improvements of the flexural properties were traced to the presence of rigid filler where the fillers will carry the major load applied to the polymer under stress condition if the interaction between the polymer and layered silicate is adequate. The fact that the polymer has a strong physisorption forces onto the stiff clay mono layers, so the soft matrix will be portion of stiff fillers, in turn, the modulus will increase. Also, the high aspect ratio of layered silicate which had high interfacial interaction within the polymer matrix can improve the modulus.

The reduction of modulus of the 5wt.% clay loading may

be related to the existence of aggregation layered silicate as well as the high number of voids in the nanocomposites structure which imparts a negative effect on the overall cross-linking properties as seen in SEM and TEM. The reason for the agglomeration could be because of the mixing parameter, i.e. the length of time for the mixing may not have been adequate, as the volume of layered silicate was increased. Moreover, as the clay loading increases, the viscosity of the nanocomposites get increased as a result; at higher clay loading, insufficient degassing will be obtained. Thus, the microvoids and porosity that are located in the structure will face difficulty in leaving the matrix system during the shear mixing. As a result, less interfacial interaction between the polymer and layered silicate will be presented[28-30] as seen in figure 4. So, there is a relationship between the experimental data and theoretical voids data. Many reports revealed a reduction in modulus due to the increment in clay loading which results in increasing the number of flaws during the nanocomposites preparation which include the voids content[16]. Thus, special care regarding the microvoids should be presented.



**Figure 4.** The relationship between the voids content and the flexural modulus

### 3.4. Tensile Modulus

**Table 4.** Tensile test outcomes for neat and different clay loading samples

Samples	Tangent modulus GPa SD %	Improvement in modulus %
Neat Vinyl ester	1.19+/- 11	0.00
Vinyl ester + 2.00 wt.% clay	1.46+/- 10	22.68
Vinyl ester + 4.00 wt.% clay	1.70+/- 4.5	42.87
Vinyl ester + 5.00 wt.% clay	1.55+/- 7	29.68

Four specimens in each loading were tested and the averages of those outcomes were obtained and are presented in Table 4. Table 4 and Figure 5 represent the enlargement of the amount of tangent modulus for nanocomposite samples compared to the pristine one. The clay loading had a strong influence on the modulus as the results were increased by the addition of nanoparticles. For example, neat vinyl ester, 2wt.%, 4 wt.% and 5 wt.% layered silicate represented 1.19, 1.46, 1.70 and 1.55GPa respectively. Again, the voids played an important role in terms of modulus since the high number of voids will reduce the interfacial interaction between the

layered silicate and polymer, in turn, less polymer physisorption forces onto the monolayer will take place[31] as depicted in figure 5.

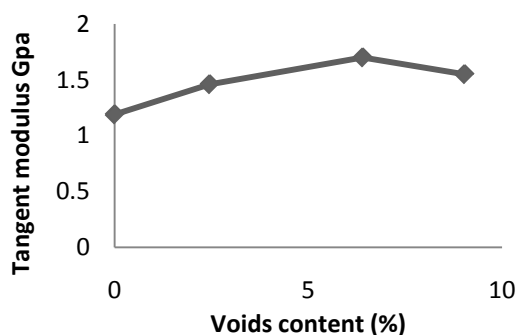


Figure 5. The influence of the voids content on the tangent modulus

## 4. Conclusions

Flexural and tensile modulus of vinyl ester matrix based on layered silicate was investigated. In addition, the effect of the percentage of microvoids in the nanocomposites structure on the flexural and tensile modulus was discussed. The layered silicate had a strong effect on the polymer properties. The higher the amount of clay loading, the better the matrix properties achieved, up to 4 wt.% clay. Further addition of clay resulted in decreasing the nanocomposites modulus which could be traced to the presence of the high amount of microvoids as calculated.

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